

# A damage-plasticity framework for fluid-infiltrating geomaterials with size-dependent anisotropy

WaiChing “Steve” Sun

Assistant Professor

Department of Civil Engineering and Engineering Mechanics,  
Fu Foundation School of Engineering and Applied Science,  
Columbia University, New York, USA.

# Acknowledgement of Current and Former Research Group Members



Kun Wang



Hyoung Suk Suh



Nikolaos Vlassis



Eric Cushman Bryant



Ran Ma, PhD



Yousef Heider,  
PhD



Chanuqi  
Liu, PhD



Jinhyun Choo,  
PhD, now HKU

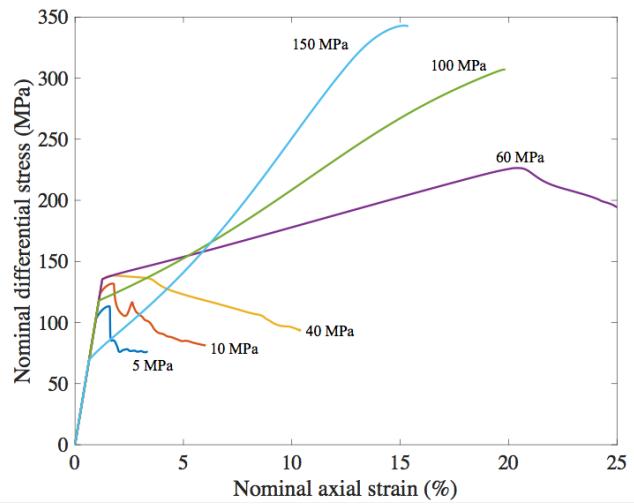
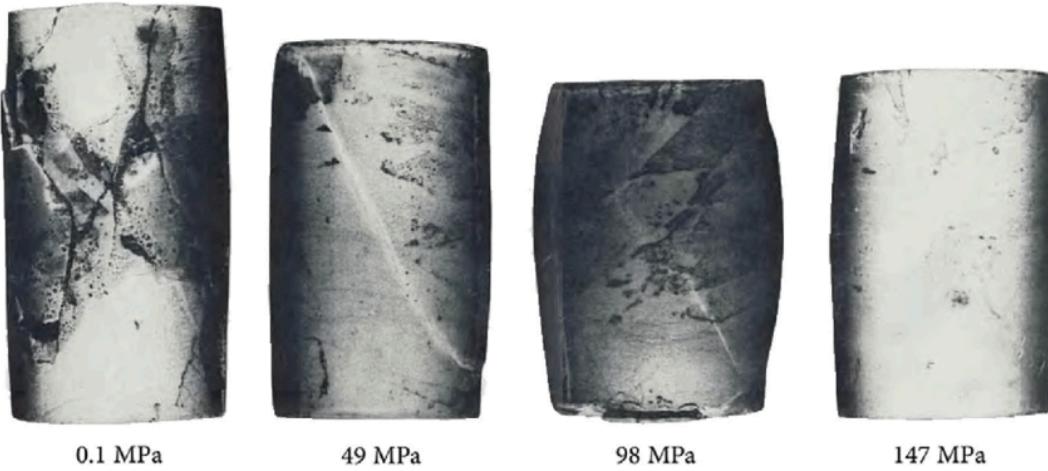


Yang Liu, PhD,  
now  
Northeastern

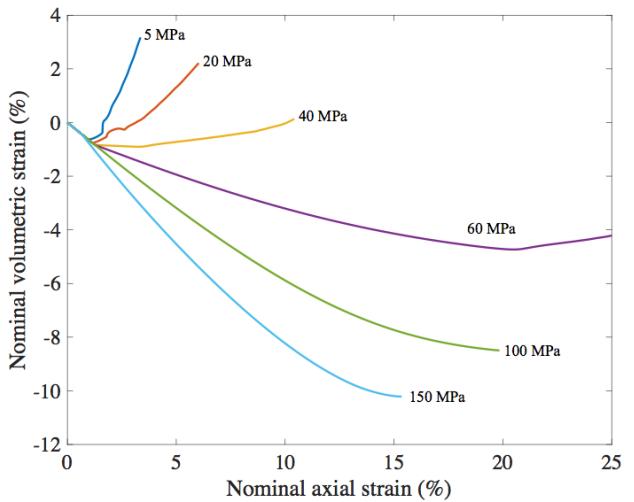


SeonHong  
Na, PhD, now  
McMaster

# Brittle-ductile transition



- Brittle regime - fracture with small scale yielding
- Quasi-brittle regime - fracture with softening
- Ductile regime - plasticity + fracture

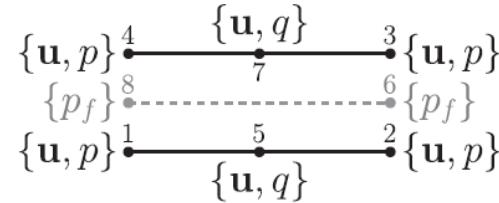


Example 1: Evolving cracks and anti-crack  
filled with pore fluid

# Background

- Common strategies
  - Cohesive zone interface element (on DG or standard FEM)  
Fracture process zone, known crack path
  - X-FEM & Assumed Strain  
Powerful and accurate approach to model problems exhibiting discontinuities and inhomogeneities, difficulty in modeling crack branching and curved crack path
- Phase-field  
Fracture surface is approximated by a phase-field.  
need fine mesh to improve convergence and geometrically represent the fracture
- Element erosion (deletion)  
Simple scheme, spurious mesh dependency
- Meshless method  
Not need to embed strong discontinuities but need to recover geometrical features via implicit function (e.g. level set)

Other possibility – peridynamic, lattice-spring, lattice-beam models



Zero-thickness interface element with hydro-mechanical coupling

e.g. Carrier, Benoit, and Sylvie Granet (2012)<sup>1</sup>

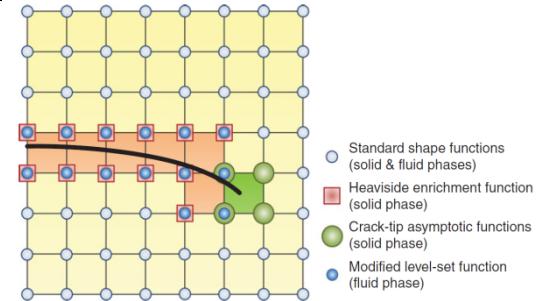
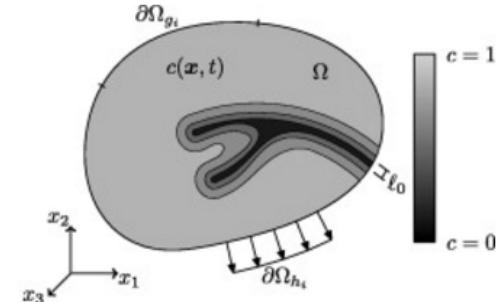


Figure 10.16 The enriched nodal points of the displacement and pressure fields in porous media

e.g. Khoei, Amir R. Extended Finite Element Method: Theory and Applications

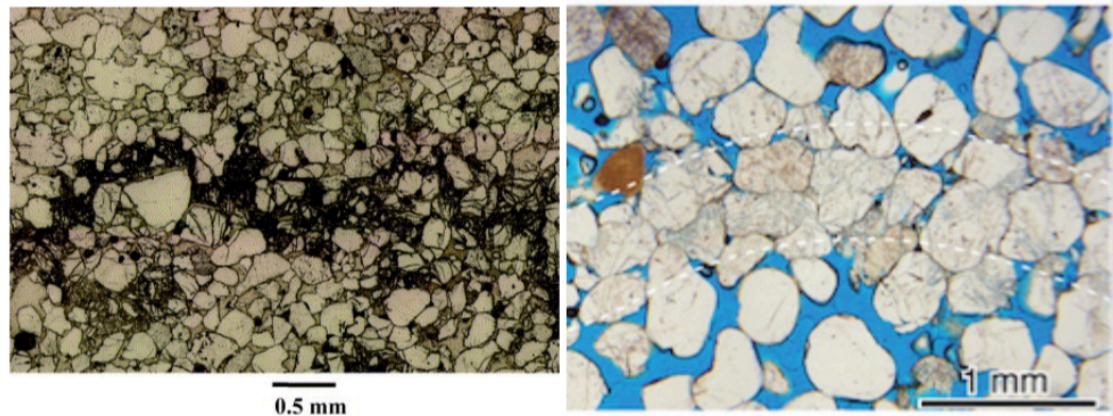


e.g. Borden, Michael J., et al. (2012)

# Compaction band as Mode I anti-crack

Compaction band in Bentheim sandstone

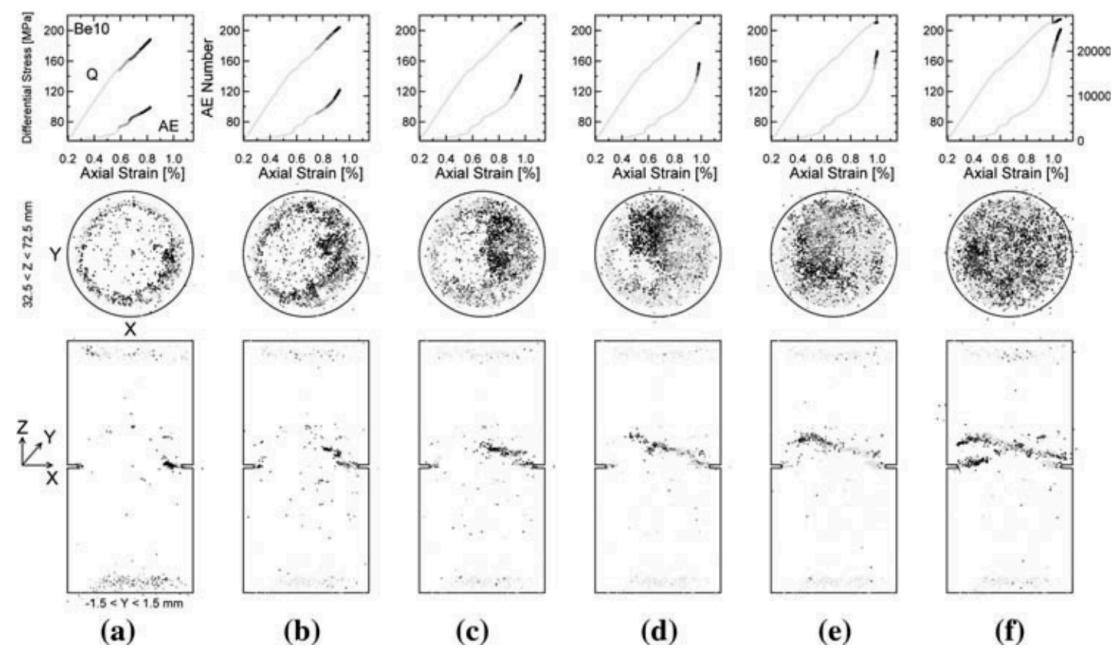
(UPPER LEFT, Baud et al. 2004,  
LOWER, Stanchits et al. 2009)



Compaction band in Navajo sandstone

(RIGHT, Fossen et al. 2011)

Compaction bands have been idealized as anti-cracks (cf. Sternolf & Pollard 2002 and Rudnicki 2007) and the associated energy release rate has been measured experimentally (e.g. Stanchits et al. 2009).



# Eigen-erosion approach

- Simple implementation

- Able to capture curved crack path, crack branching, coalescence
- Mesh independent
- Robust

- Eigen-fracture ( Schmidt et al. 2009 )

- Eigen-fracture energy functional:

$$F_\epsilon(u, \varepsilon^*, t) = \int_{\Omega} W(\varepsilon(u) - \varepsilon^*) \, dV - \int_{\Gamma_2} \bar{T} \cdot u \, dS + G_c \frac{|C_\epsilon|}{2\epsilon}$$

- Phase field energy functional:

$$F_\epsilon(u, \varphi, t) = \int_{\Omega} W(\varepsilon(u), \varphi) \, dV - \int_{\Gamma_2} \bar{T} \cdot u \, dS + \int_{\Omega} G_c \left( \frac{1}{2\epsilon} \|1 - \varphi\|^2 + \frac{\epsilon}{2} \|\nabla \varphi\|^2 \right) \, dV$$

- Eigen-erosion ( Pandolfi and Ortiz. 2012 )

- Restrict Eigen-deformation to binary case: either intact or completely failed
- For FEM: leads to element erosion scheme that is mesh-independent

Regularized fracture  
energy term

# Formulation of variational eigen-erosion in fluid-saturated porous media

Free Energy functionals

$$W(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*, p^f) = W_{\text{eff}}(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*) + W_{\text{fluid}}(\boldsymbol{\varepsilon}, p^f)$$

$$W_{\text{eff}}(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*) = \int_{\Omega} \frac{1}{2} (\boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^*) : \mathcal{C}^e : (\boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^*) dV \quad W_{\text{fluid}}(\boldsymbol{\varepsilon}, p^f) = \frac{M}{2} [(1-b)\varepsilon_v - \frac{p^f}{M}]^2$$

with eigen-deformation corresponding to fracture (set C) and compaction band (set CB)

$$\boldsymbol{\varepsilon}^*(\boldsymbol{x}, t) = \begin{cases} \boldsymbol{\varepsilon}(\boldsymbol{x}, t) & \boldsymbol{x} \in C \\ \varepsilon_{\text{inel}} \boldsymbol{n}_{CB}(\boldsymbol{x}, t) \otimes \boldsymbol{n}_{CB}(\boldsymbol{x}, t) & \boldsymbol{x} \in CB \\ \mathbf{0} & \boldsymbol{x} \in \Omega \setminus (C \cup CB) \end{cases}$$

C: Crack set  
 CB: Compaction Band set  
 Intact element

Fluid dissipation

$$\mathcal{D}_f(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*, p^f) = - \int_0^t \left[ \int_{\Omega} \frac{1}{2} \frac{\boldsymbol{\kappa}(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*, p^f)}{\mu} : (\nabla p^f \otimes \nabla p^f) dV \right] d\tau$$

Enhanced or suppressed permeability due to fracture and compaction band

# Variational Statement of the variational eigen-erosion in fluid-saturated porous media

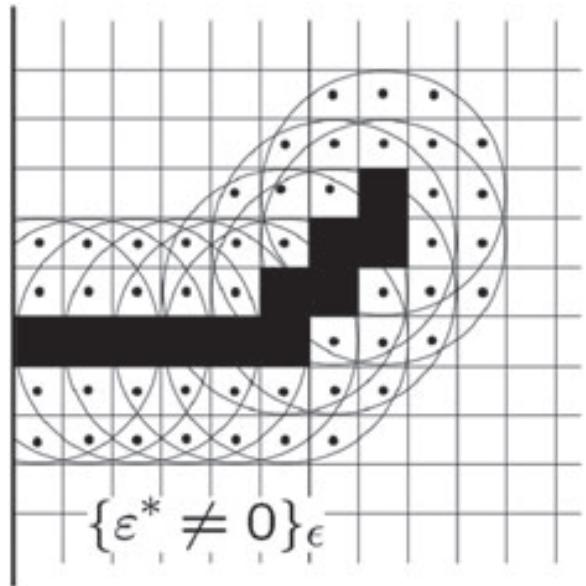
Find :the saddle point of the Energy-dissipation Functional with respect to solid displacement and pore pressure

$$\begin{aligned}
 F_{(\epsilon_C, \epsilon_{CB})}(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*, p^f, t) = & \int_{\Omega} \frac{1}{2} (\boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^*) : \mathcal{C}^e : (\boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^*) dV - \int_{\Gamma_t} \bar{\mathbf{T}} \cdot \mathbf{u} dS \\
 & + \int_{\Omega} \frac{M}{2} [(1-b)\boldsymbol{\varepsilon}_v - \frac{p^f}{M}]^2 dV + \int_0^t [\int_{\Omega} \bar{\mathbf{s}} \cdot \mathbf{p}^f dV - \int_{\Gamma_q} \bar{\mathbf{q}} \cdot \mathbf{p}^f dS] d\tau \\
 & + G_C \frac{|C_{\epsilon_C}|}{2\epsilon_C} + G_{CB} \frac{|C_{\epsilon_{CB}}|}{2\epsilon_{CB}} + \mathcal{D}_f(\boldsymbol{\varepsilon}, \boldsymbol{\varepsilon}^*, p^f).
 \end{aligned}$$

Subjected to a monotonicity constraint that enforces history-dependence

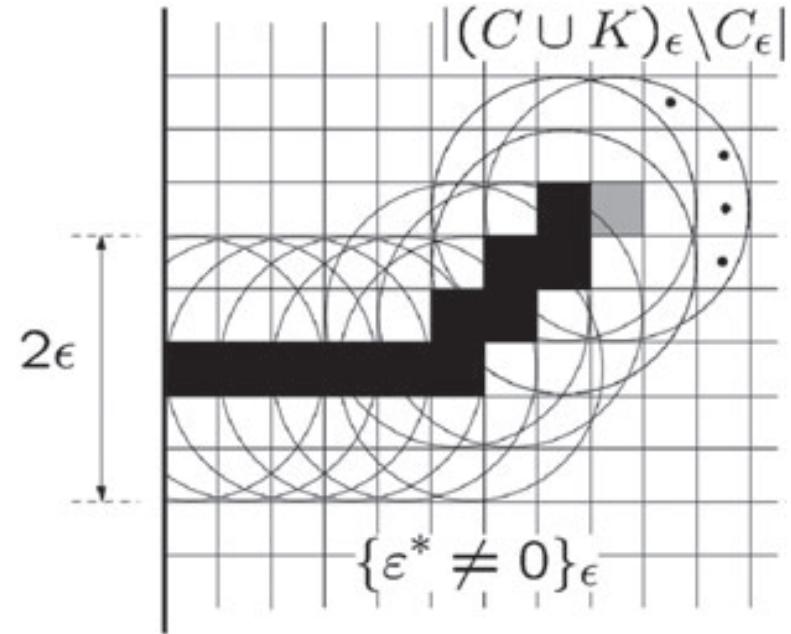
$$\left\{ \begin{array}{l} C(t) \subset C(t + \Delta t) \\ CB(t) \subset CB(t + \Delta t) \end{array} \right. \quad \boldsymbol{\varepsilon}^*(\mathbf{x}, t) = \begin{cases} \boldsymbol{\varepsilon}(\mathbf{x}, t) & \mathbf{x} \in C \\ \boldsymbol{\varepsilon}_{\text{inel}} \mathbf{n}_{CB}(\mathbf{x}, t) \otimes \mathbf{n}_{CB}(\mathbf{x}, t) & \mathbf{x} \in CB \\ \mathbf{0} & \mathbf{x} \in \Omega \setminus (C \cup CB) \end{cases}$$

# Departure from classical element deletion method – Non-locality from crack neighborhood



Crack set and intact  
element in epsilon  
neighborhood

Crack area increment

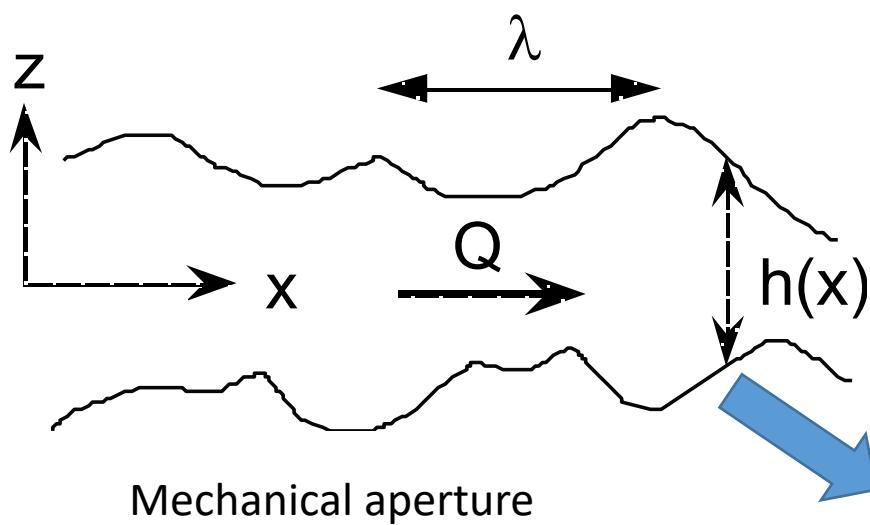


Crack set in the epsilon  
neighborhood

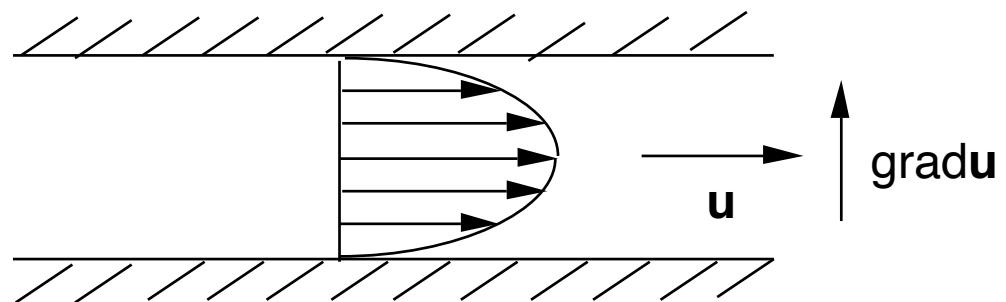
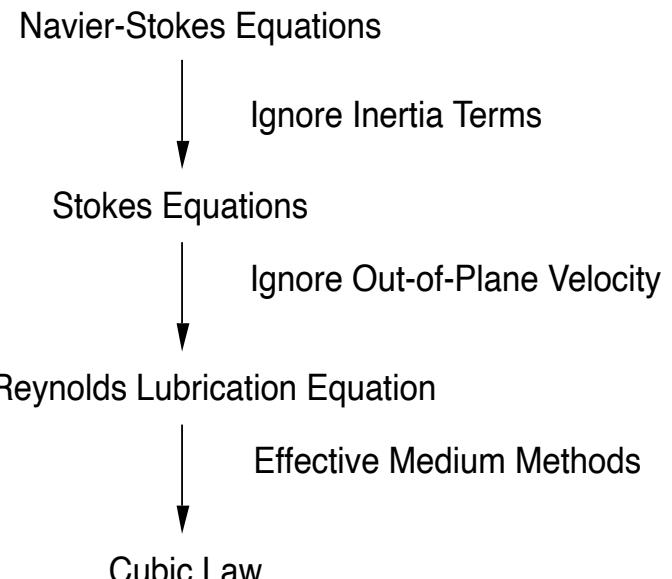
$$\Delta A_K = \frac{|(\mathcal{C} \cup \mathcal{K})_\epsilon \setminus \mathcal{C}_\epsilon|}{2\epsilon}$$

Pandolfi, Anna, and Michael Ortiz. (2012)

# Mechanical vs. Hydraulic Aperture



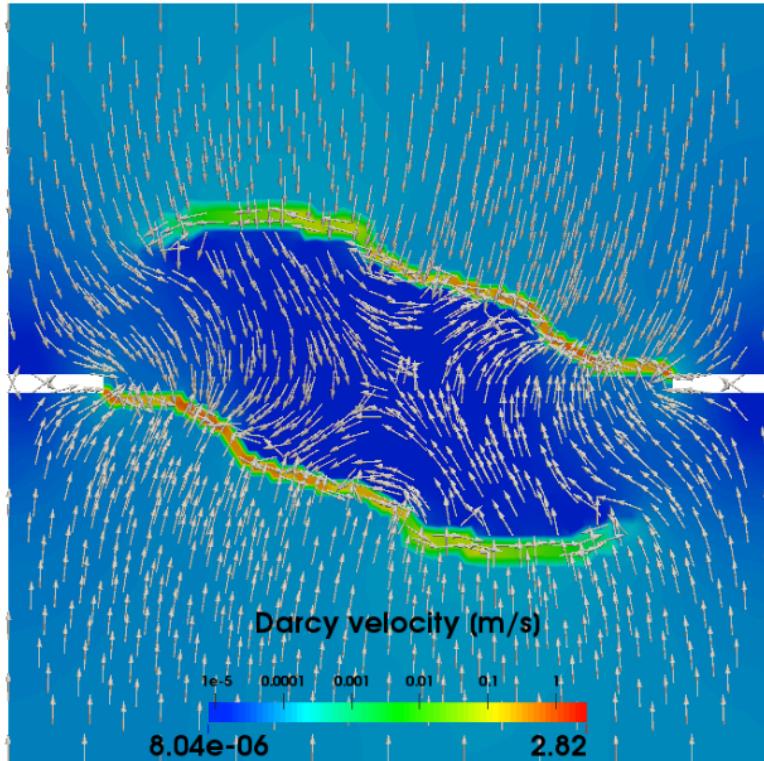
Mechanical aperture



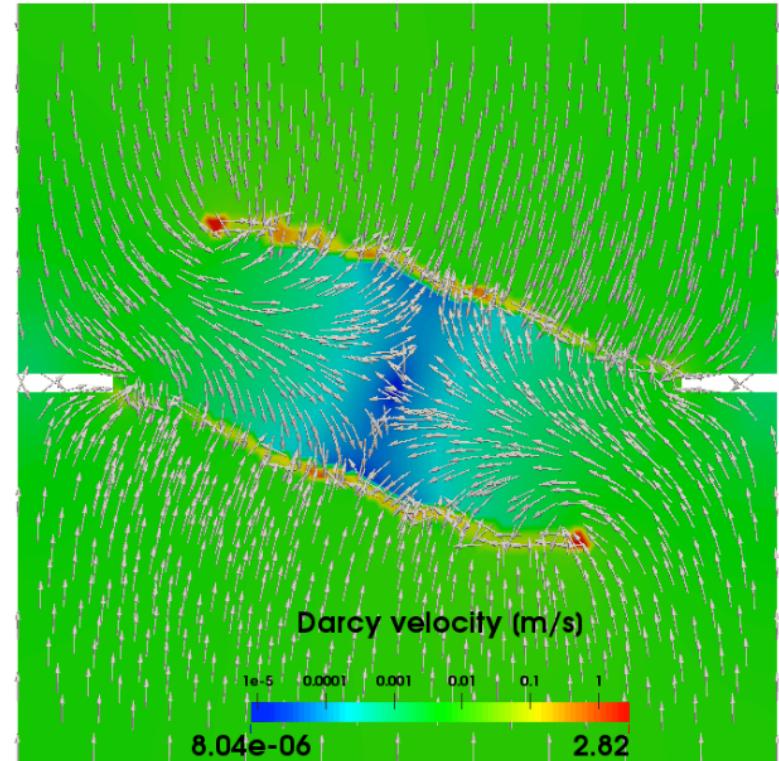
Hydraulic aperture

Figures from Robert Zimmerman

# Effect of flow conduit created by cracks



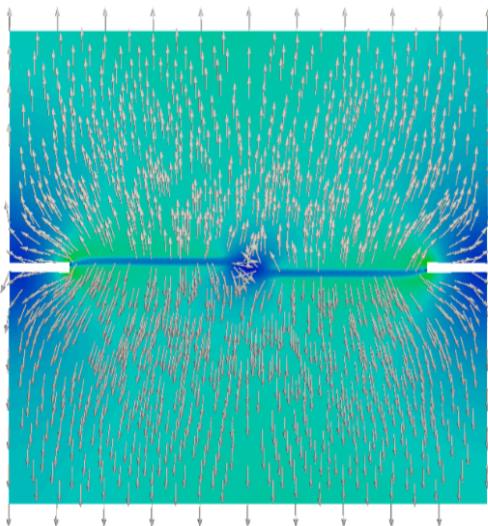
strain rate  $0.0005 \text{ s}^{-1}$



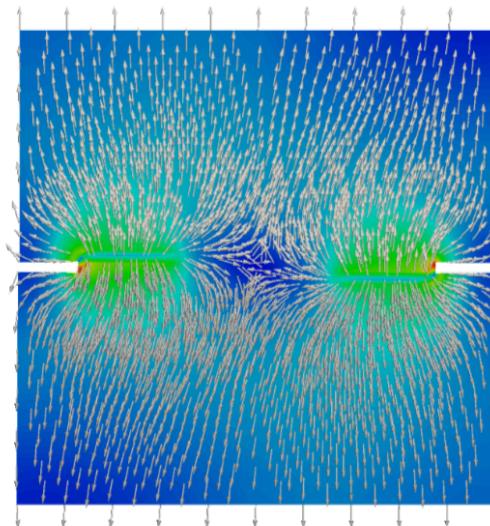
strain rate  $0.05 \text{ s}^{-1}$

Test 1: Shear loading  $P_s = 50\text{MPa}$ ,  $0.05\text{s}^{-1}$   
Vertical strain rate =  $0.0005\text{s}^{-1}$

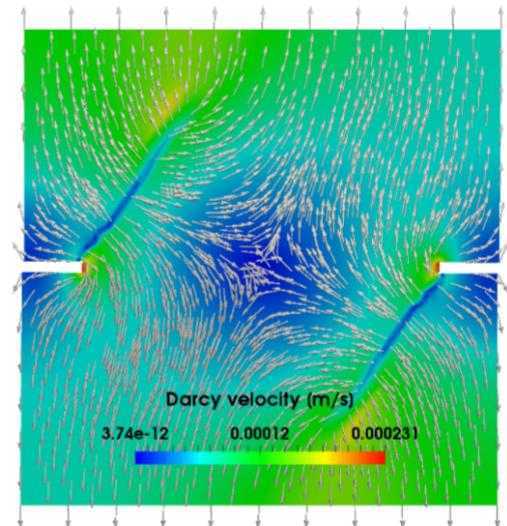
# Example: Compaction band as flow Barrier



(a)  $P_s = 50 \text{ MPa}$



(b)  $P_s = 100 \text{ MPa}$



(c)  $P_s = 120 \text{ MPa}$

# Example: Crack propagation and coalescence in heterogeneous porous media

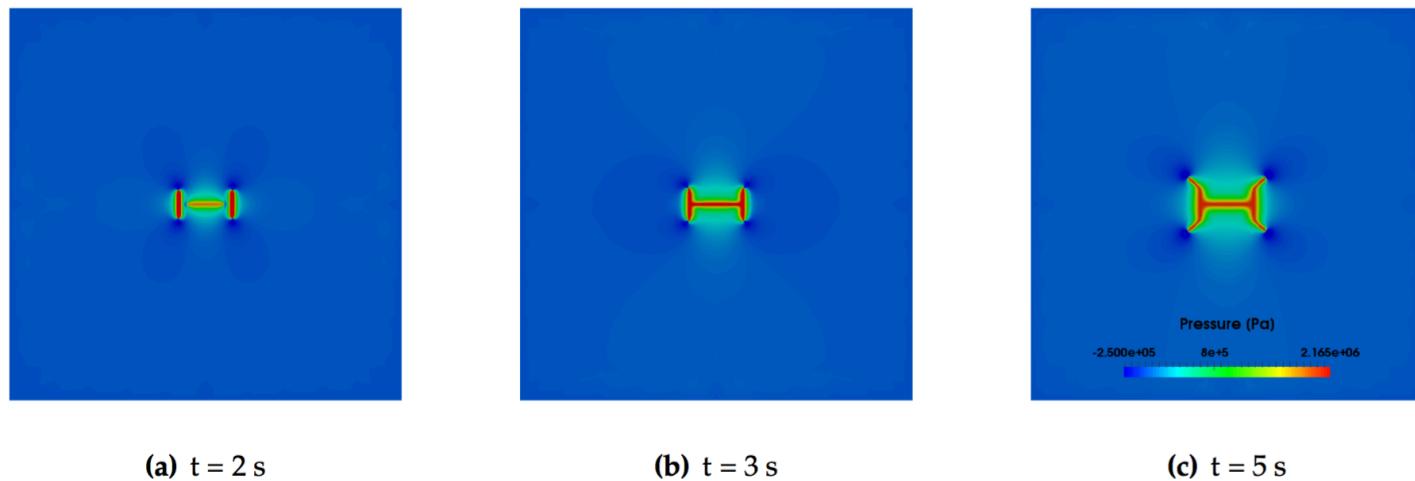
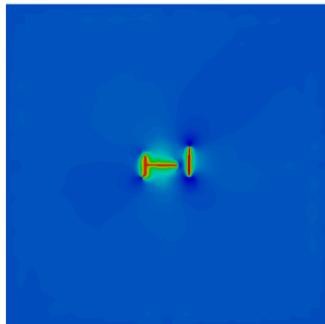


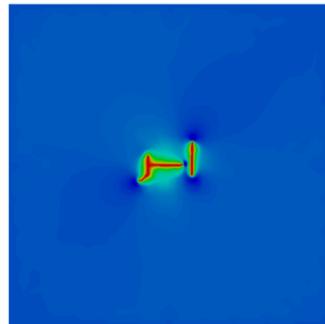
Fig. 21: Fluid driven propagation and coalescence of fractures in homogeneous porosity field.  $v_{\text{inject}} = 0.00025\text{ m/s}$ . Fluid pressure at  $t = 2\text{ s}, 3\text{ s}, 5\text{ s}$ .

Wang & Sun, CMAME, 2017.  
Phase field version: Wick & Wheeler, SPE, 2014

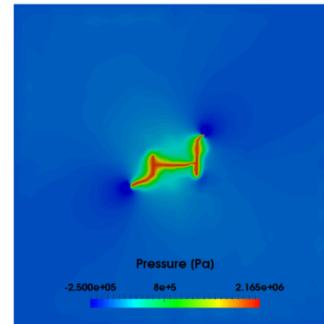
# Example: Effect of injection rate in heterogeneous porous media



(a)  $t = 2\text{ s}$

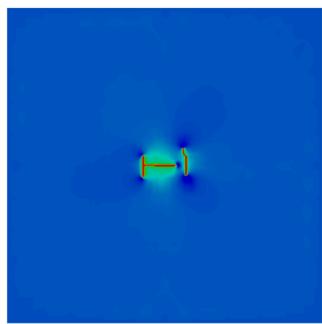


(b)  $t = 3\text{ s}$

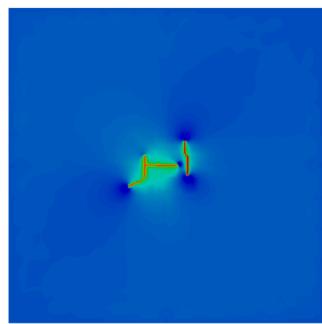


(c)  $t = 5\text{ s}$

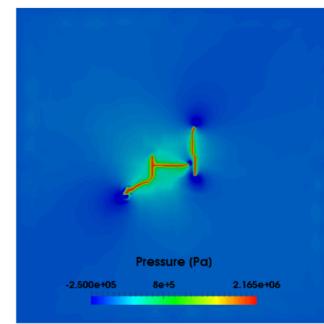
Fig. 22: Fluid driven propagation and coalescence of fractures in heterogeneous porosity field.  $v_{\text{inject}} = 0.00025\text{ m/s}$ . Fluid pressure at  $t = 2\text{ s}, 3\text{ s}, 5\text{ s}$ .



(a)  $t = 0.2\text{ s}$



(b)  $t = 0.3\text{ s}$



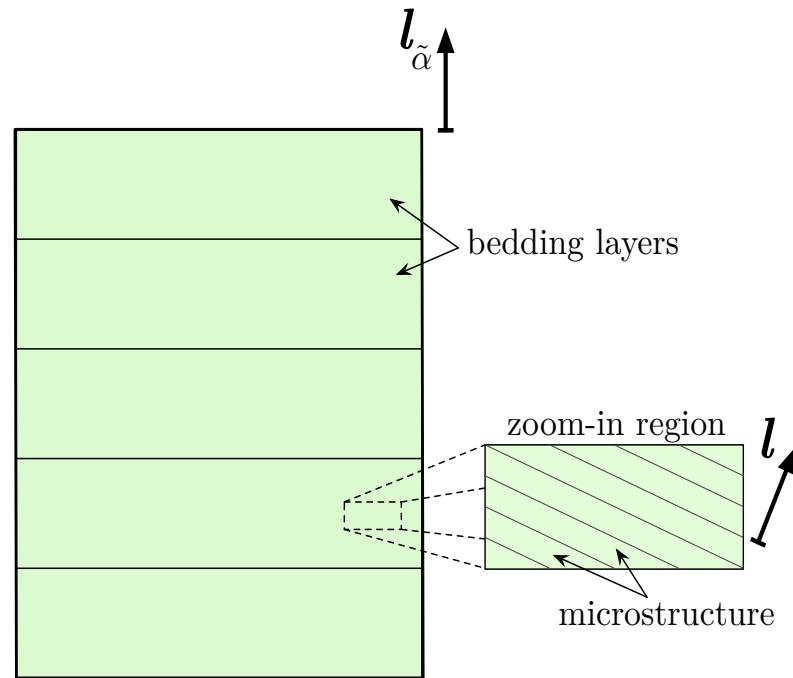
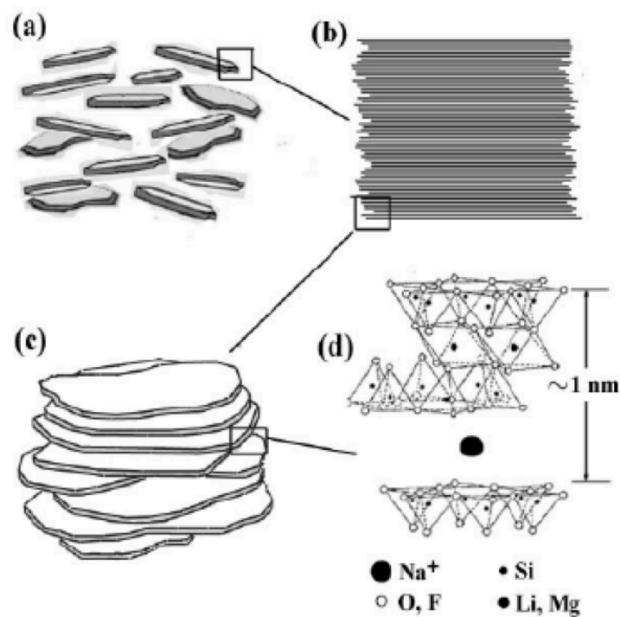
(c)  $t = 0.5\text{ s}$

Fig. 23: Fluid driven propagation and coalescence of fractures in heterogeneous porosity field.  $v_{\text{inject}} = 0.0025\text{ m/s}$ . Fluid pressure at  $t = 0.2\text{ s}, 0.3\text{ s}, 0.5\text{ s}$ .

Wang & Sun,  
CMAME 2017.

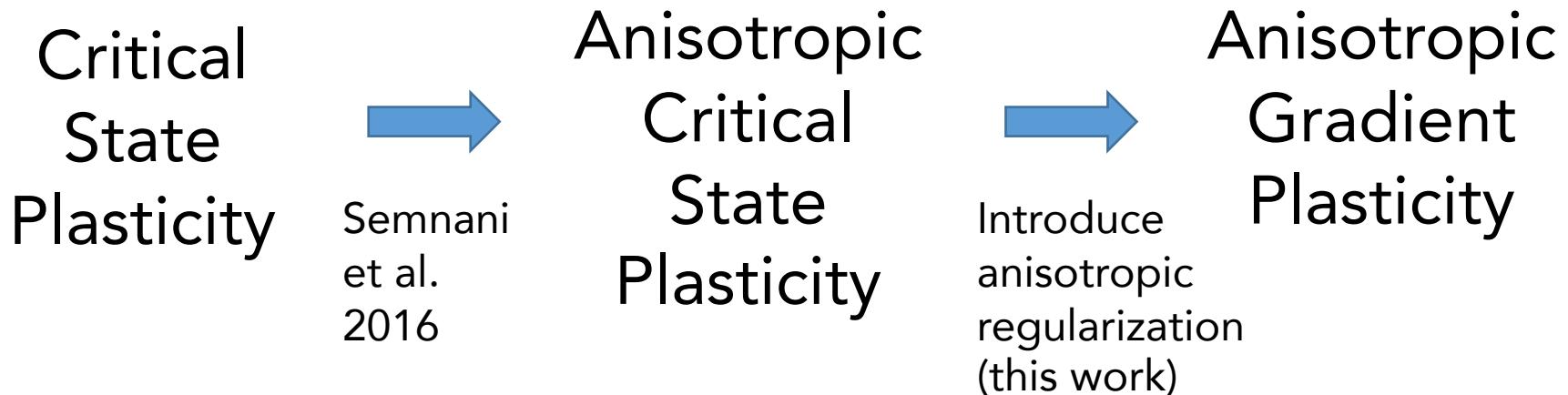
Anisotropic materials in the ductile regime

# Motivation



# Size-dependent anisotropy

- Goal: Introduce different “effective” anisotropy or different size of specimen numerically
- Strategy:



# Review on Anisotropic Cam-clay model

## Anisotropic Elasticity

$$W(\boldsymbol{\epsilon}^e, \boldsymbol{\alpha}, \tilde{\boldsymbol{\alpha}}) = W^e(\boldsymbol{\epsilon}^e) + W^p(\boldsymbol{\alpha}, \tilde{\boldsymbol{\alpha}}).$$

where  $W^e = \frac{1}{2}\boldsymbol{\epsilon}^e : \mathbf{C}^e : \boldsymbol{\epsilon}^e$ . and  $\mathbf{C}^e = c_1\mathbf{E}_1 + c_2\mathbf{E}_2 + c_3(\mathbf{E}_3 + \mathbf{E}_4) + c_5\mathbf{F} + c_6\mathbf{G}$

The key idea is to introduce  $p^*$  and  $q^*$  as weighted norms to simplify implementation

$$p^* = \mathbf{a}_p : \boldsymbol{\sigma}, \quad q^* = \sqrt{\frac{1}{2}\boldsymbol{\sigma} : \mathbf{A}_q : \boldsymbol{\sigma}},$$

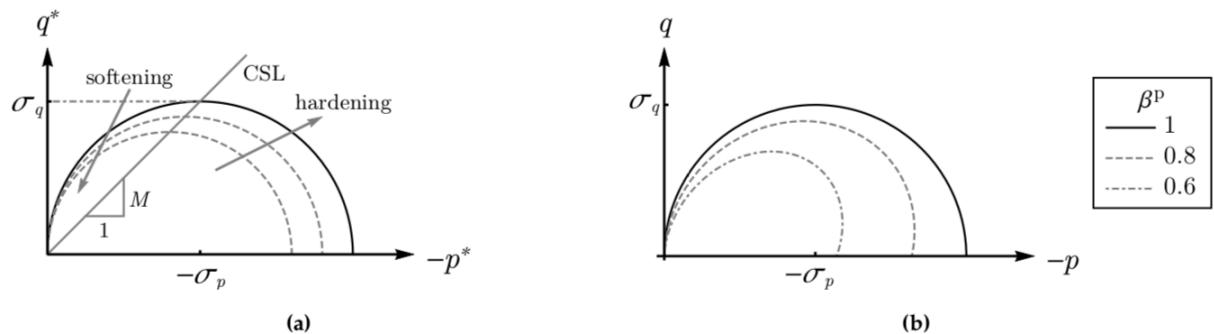


Fig. 1: Geometric interpretation of the mapped two-invariant yield criterion, Eq. (34): (a) introducing  $M$  as the slope of the critical state line (CSL) in the starred scalar space, after [Ortiz and Pandolfi \[2004\]](#), and with softening and hardening trends after [Borja \[2013\]](#); and, (b) unmapped two-invariant yield criterion parameterized by the mapping coefficient, varying  $\beta^{\text{dev}} = \beta^{\text{vol}}$  at fixed  $\alpha^{\text{dev}} = \alpha^{\text{vol}} = \gamma^{\text{dev}} = \gamma^{\text{vol}} = 1$ , after [Semnani et al. \[2016\]](#).

# Anisotropic gradient regularization

---

Two additional energy functionals are added to introduce gradient dependence on the plastic work

$$W_{\tilde{\alpha}}^p(\boldsymbol{\alpha}, \tilde{\boldsymbol{\alpha}}) = \overbrace{\frac{k_\lambda}{2} (\tilde{\lambda} - \lambda)^2 + \frac{k_v}{2} (\tilde{\epsilon}_v^p - \epsilon_v^p)^2}^{\text{penalty functional terms}} + \overbrace{\frac{K_\lambda l_\lambda^2}{2} \nabla \tilde{\lambda} \cdot \boldsymbol{\omega}_\lambda \cdot \nabla \tilde{\lambda} + \frac{K_v l_v^2}{2} \nabla \tilde{\epsilon}_v^p \cdot \boldsymbol{\omega}_v \cdot \nabla \tilde{\epsilon}_v^p}^{\text{regularization functional terms}},$$

where the structural tensor for the **anisotropic regularization** is

$$\boldsymbol{\omega}_\lambda = (\mathbf{p}_\lambda \otimes \mathbf{p}_\lambda) : \mathbf{1} = \mathbf{1} + \phi_\lambda \boldsymbol{\phi}_\lambda + \chi_\lambda \boldsymbol{\chi}_\lambda, \quad \mathbf{p}_\lambda = \boldsymbol{\phi}_\lambda \sqrt{1 + \phi_\lambda} + \boldsymbol{\chi}_\lambda \sqrt{1 + \chi_\lambda},$$

# Anisotropic gradient regularization

Incremental update are obtained from Euler-Lagrange equation of the incremental energy functional, i.e.,

$$\zeta_{n+1} = \arg \min_{\zeta_{n+1}} W(\zeta_{n+1}), \quad \zeta_{n+1} = \{\epsilon_{n+1}^e, \tilde{\lambda}_{n+1}, \tilde{\epsilon}_v^p\},$$

which leads to

Local governing  
equation for  
constitutive laws

$$\frac{\delta W(\zeta_{n+1})}{\delta \epsilon_{n+1}^e} = \underbrace{\sigma_{n+1} - \sigma_q|_{n+1} \frac{\partial \Delta \lambda}{\partial \Delta \epsilon^p} - \sigma_p|_{n+1} b_v}_{\text{purely local terms}} + \underbrace{k_\lambda (\tilde{\lambda}_{n+1} - \lambda_{n+1}) \frac{\partial \Delta \lambda}{\partial \Delta \epsilon^p} + k_v (\tilde{\epsilon}_v^p|_{n+1} - \epsilon_v^p|_{n+1}) b_v}_{\text{micromorphic terms}} = \mathbf{0},$$

Regularized modified  
Helmholtz equations

$$\frac{\delta W(\zeta_{n+1})}{\delta \tilde{\lambda}_{n+1}} = k_\lambda (\tilde{\lambda}_{n+1} - \lambda_{n+1}) - K_\lambda l_\lambda^2 \nabla \cdot (\boldsymbol{\omega}_\lambda \cdot \nabla \tilde{\lambda}_{n+1}) = 0, \text{ in } \mathcal{B},$$
$$\frac{\delta W(\zeta_{n+1})}{\delta \tilde{\epsilon}_v^p|_{n+1}} = k_v (\tilde{\epsilon}_v^p|_{n+1} - \epsilon_v^p|_{n+1}) - K_v l_v^2 \nabla \cdot (\boldsymbol{\omega}_v \cdot \nabla \tilde{\epsilon}_v^p|_{n+1}) = 0, \text{ in } \mathcal{B},$$

# Special case: Isotropic gradient regularization

- Nonlocal Modified Cam-Clay (MCC) model

- Yield function (cf. Borja, 2013)

$$f_y = \frac{q^2}{M^2} + p'(p' - p'_c) = 0$$

where  $p' = \frac{1}{r} \text{tr}(\boldsymbol{\sigma}')$ ,  $q = \sqrt{\frac{2}{3}} \| \mathbf{s} \|$ ,  $\mathbf{s} = \boldsymbol{\sigma}' - p'\mathbf{1}$

- Hardening

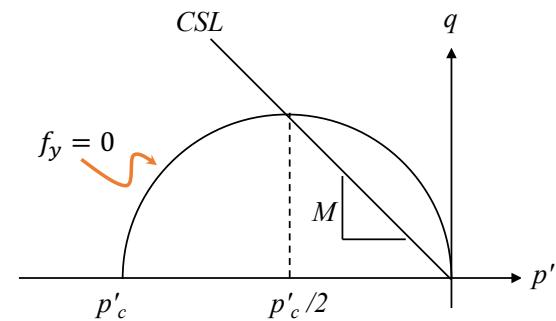
$$\dot{p}'_c = -\frac{\dot{\varepsilon}_v^p}{c_c - c_r} p'_c$$

(using the bilogarithmic relation between the specific volume and  $p'_c$ )

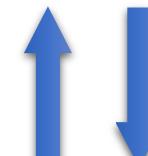
- The discrete hardening relation in a residual form

$$f_r = p'_c - p'_{c,old} \exp\left(\frac{\varepsilon_v^e - \varepsilon_v^{e \text{ trial}}}{c_c - c_r}\right) + \epsilon_p (\bar{\alpha} - \alpha) = 0$$

$\epsilon_p$  : penalty parameter  
 $\bar{\alpha} \equiv p'_c$  : local internal variable  
 $\alpha$  : global micromorphic variable



**Global Iteration**



**Local Iteration**

$$\begin{cases} \nabla \cdot (\boldsymbol{\sigma}' - p\mathbf{1}) + \rho\mathbf{g} = 0 \\ \nabla \cdot \dot{\mathbf{u}} + \nabla \cdot \left(-\frac{1}{\rho_f g} \mathbf{k} \cdot \nabla p\right) = 0 \\ \alpha - l_p^2 \Delta \alpha = \bar{\alpha} \end{cases}$$

$$\mathbf{r} = \begin{Bmatrix} \varepsilon_v^e - \varepsilon_v^{e \text{ trial}} + \Delta \lambda \partial_{p'} f \\ \varepsilon_s^e - \varepsilon_s^{e \text{ trial}} + \Delta \lambda \partial_{q'} f \\ f_y \\ f_r \end{Bmatrix}; \quad \mathbf{x} = \begin{Bmatrix} \varepsilon_v^e \\ \varepsilon_s^e \\ \Delta \lambda \\ p'_c \end{Bmatrix}$$

# Anisotropic gradient regularization

Incremental update are obtained from Euler-Lagrange equation of the incremental energy functional, i.e.,

$$\zeta_{n+1} = \arg \min_{\zeta_{n+1}} W(\zeta_{n+1}), \quad \zeta_{n+1} = \{\epsilon_{n+1}^e, \tilde{\lambda}_{n+1}, \tilde{\epsilon}_v^p\},$$

which leads to

Local governing  
equation for  
constitutive laws

$$\frac{\delta W(\zeta_{n+1})}{\delta \epsilon_{n+1}^e} = \underbrace{\sigma_{n+1} - \sigma_q|_{n+1} \frac{\partial \Delta \lambda}{\partial \Delta \epsilon^p} - \sigma_p|_{n+1} b_v}_{\text{purely local terms}} + \underbrace{k_\lambda (\tilde{\lambda}_{n+1} - \lambda_{n+1}) \frac{\partial \Delta \lambda}{\partial \Delta \epsilon^p} + k_v (\tilde{\epsilon}_v^p|_{n+1} - \epsilon_v^p|_{n+1}) b_v}_{\text{micromorphic terms}} = \mathbf{0},$$

Regularized modified  
Helmholtz equations

$$\frac{\delta W(\zeta_{n+1})}{\delta \tilde{\lambda}_{n+1}} = k_\lambda (\tilde{\lambda}_{n+1} - \lambda_{n+1}) - K_\lambda l_\lambda^2 \nabla \cdot (\boldsymbol{\omega}_\lambda \cdot \nabla \tilde{\lambda}_{n+1}) = 0, \text{ in } \mathcal{B},$$
$$\frac{\delta W(\zeta_{n+1})}{\delta \tilde{\epsilon}_v^p|_{n+1}} = k_v (\tilde{\epsilon}_v^p|_{n+1} - \epsilon_v^p|_{n+1}) - K_v l_v^2 \nabla \cdot (\boldsymbol{\omega}_v \cdot \nabla \tilde{\epsilon}_v^p|_{n+1}) = 0, \text{ in } \mathcal{B},$$

# Plastic Flow direction w/ increasing anisotropic regularization effect

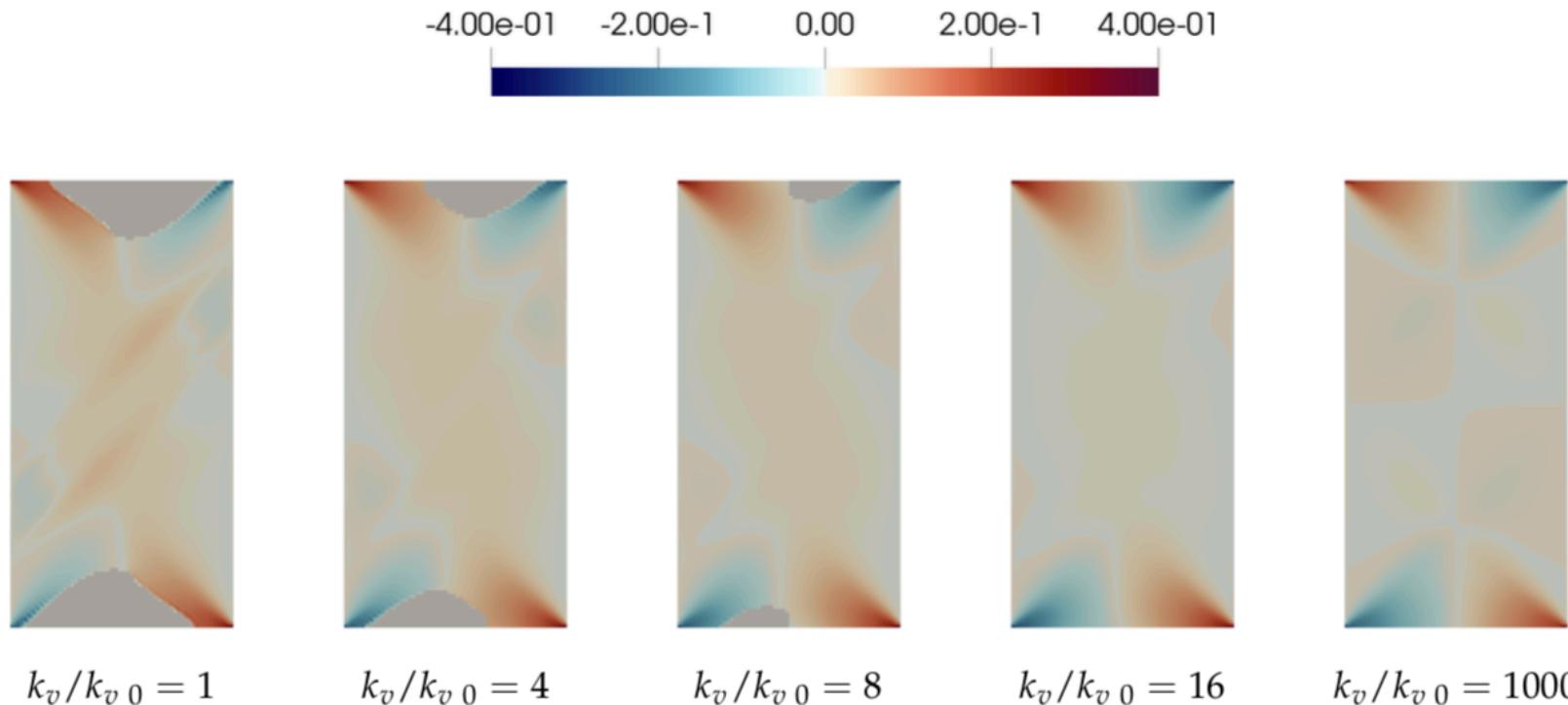


Fig. 12: Angle from horizontal of plastic strain's eigenvector corresponding to its greatest (most dilative) in-plane eigenvalue (in radians), for diffusively anisotropic plane strain compression case, with  $\phi_v = 16$  and  $l_v$  described by  $\theta_v = \pi/4$ , at  $u_2 = -1.0 \times 10^{-3}$  mm. Regions where  $\epsilon^P = 0$  are grayed-out (LI-MA).

Lesson. We can introduce plastic flow non-coaxial to the stress gradient of the yield function through anisotropic regularization!

# Strain localization w/ increasing anisotropic regularization effect

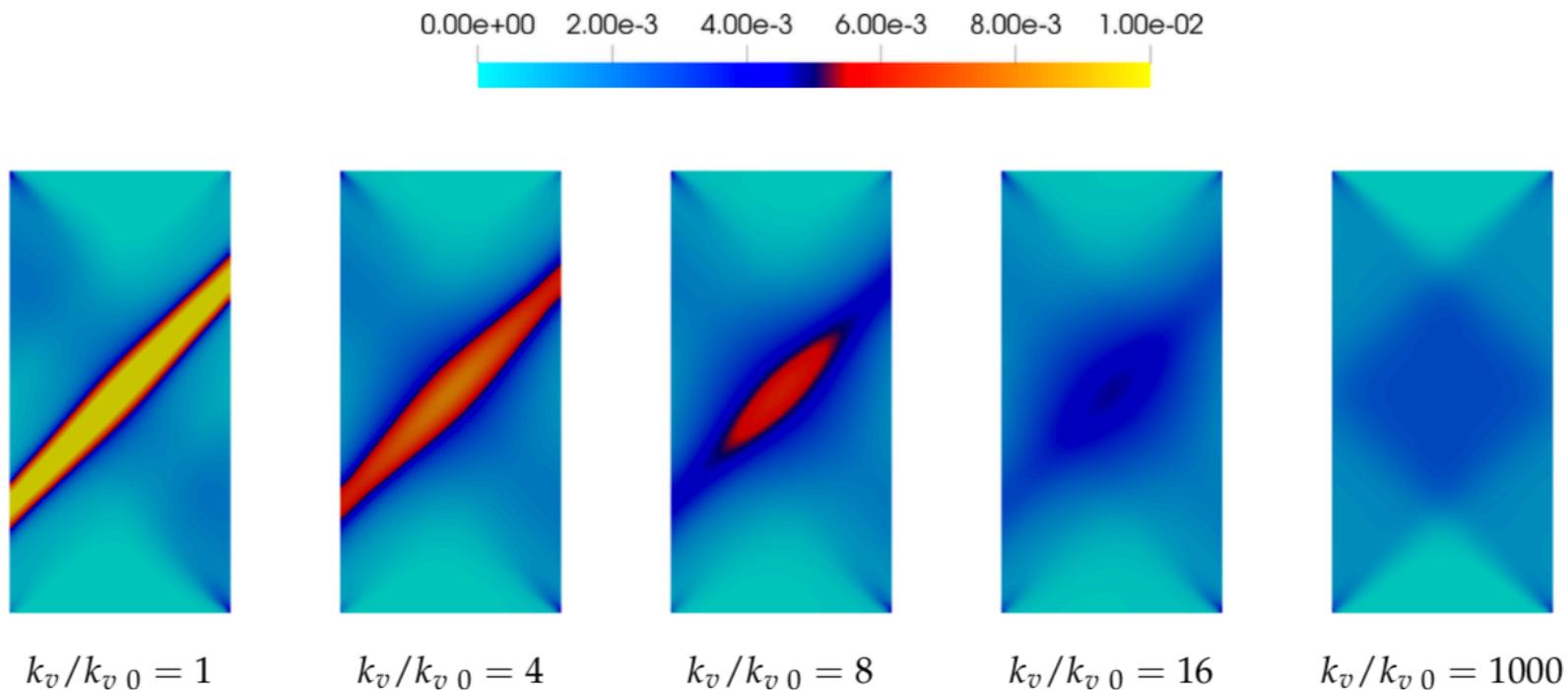
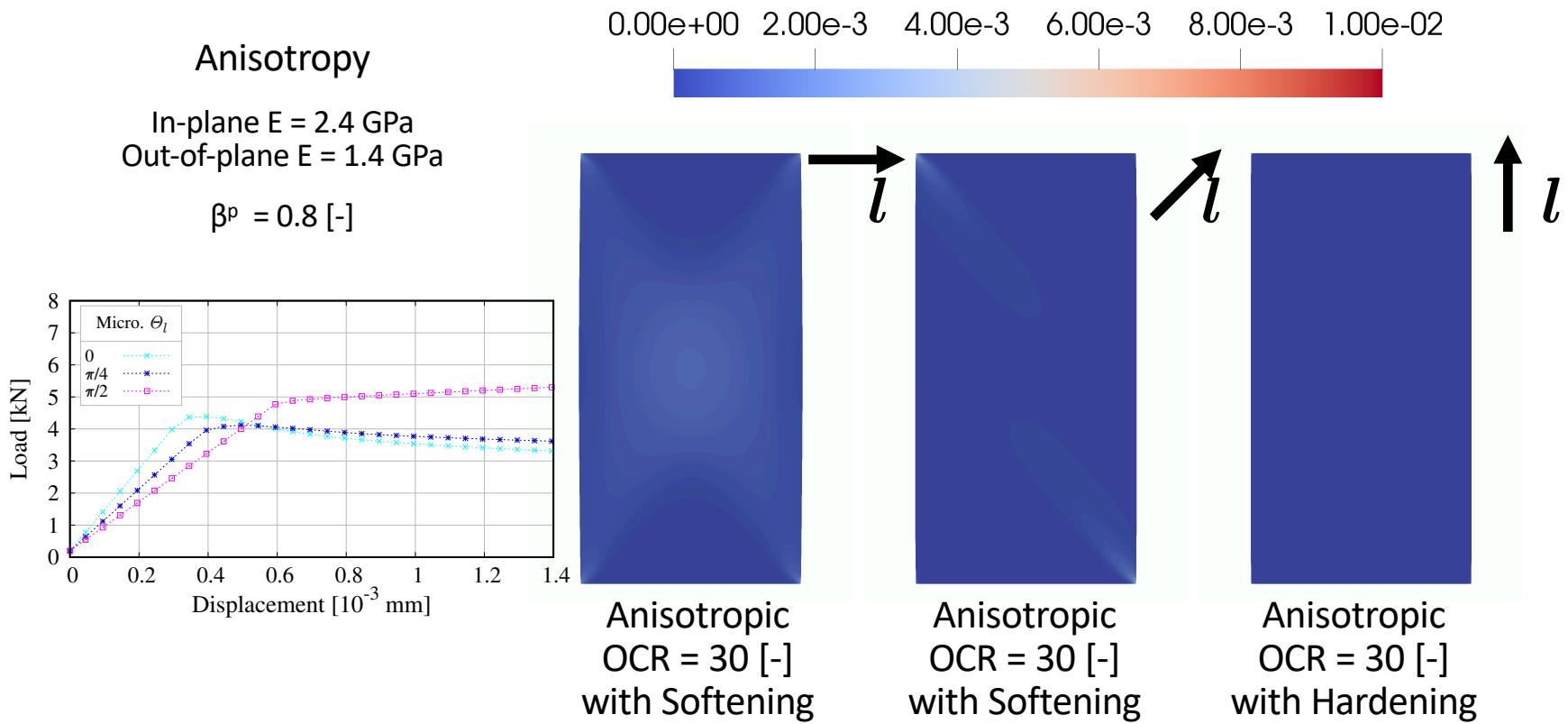
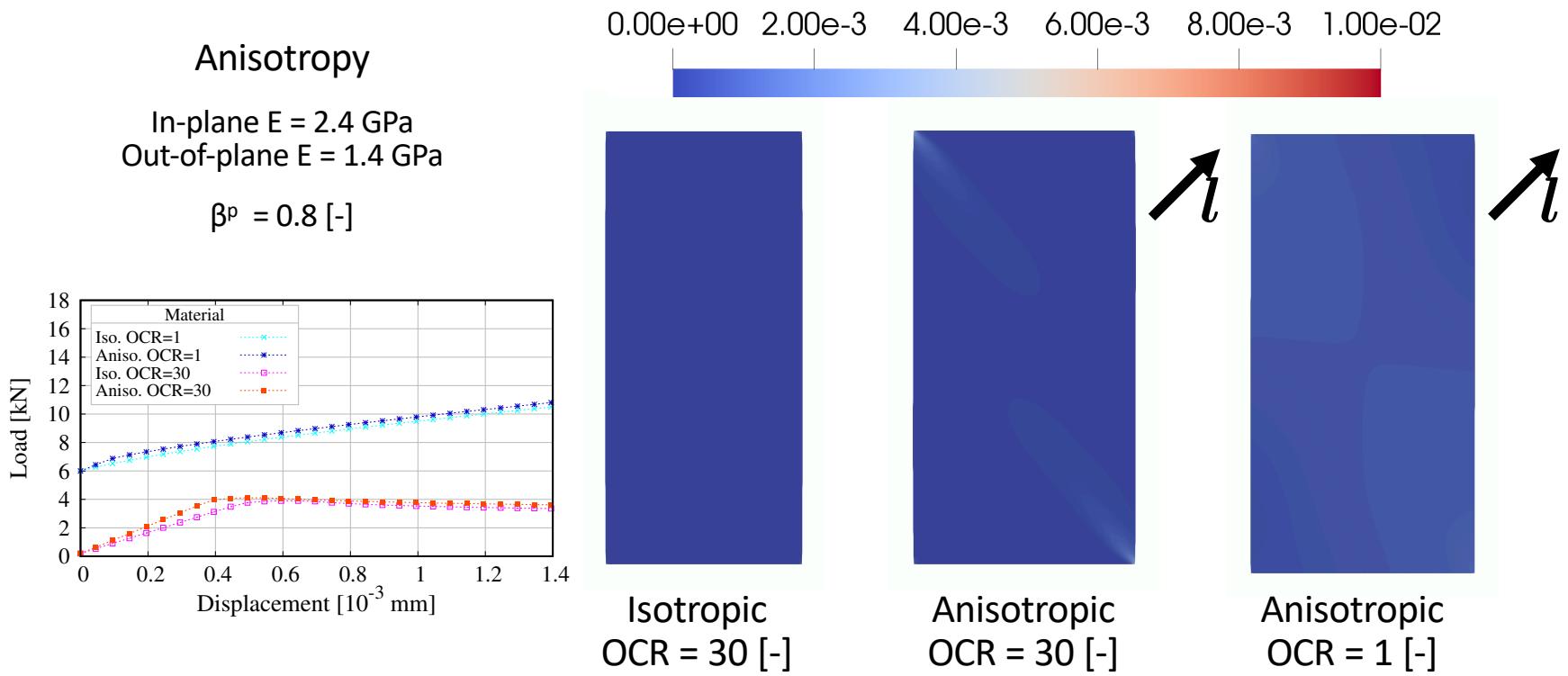


Fig. 11: Equivalent plastic strain  $\lambda$  for diffusively anisotropic plane strain compression case, with  $\phi_v = 16$  and  $l_v$  described by  $\theta_v = \pi/4$ , at  $u_2 = -1.0 \times 10^{-3}$  mm (LI-MA).

# Boundary Value Problems: Rotation of Microstructural Direction

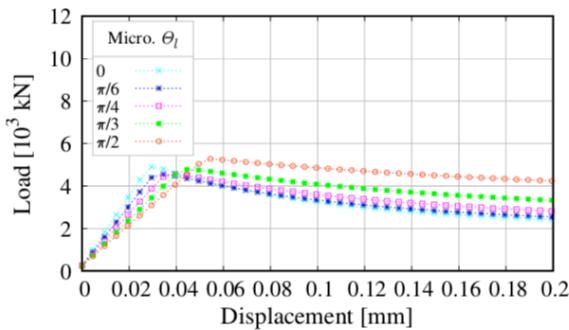
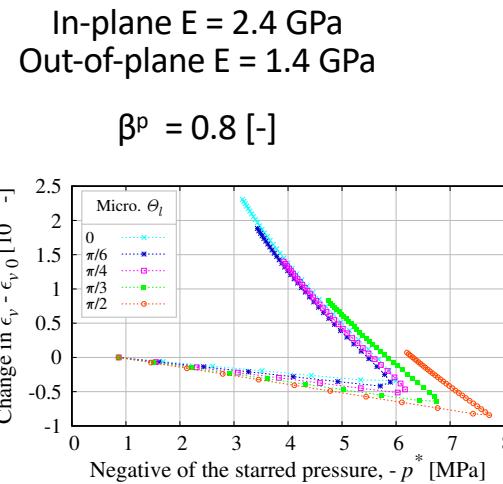


# Micromorphic Regularization of Boundary Value Problems: Isotropic vs. Anisotropic



# Boundary Value Problems: 3D Rotation of Microstructural Direction

## Anisotropy



Anisotropic  
OCR = 30 [-]  
with Softening



Anisotropic  
OCR = 30 [-]  
with Diffuse Softening

# Coupled damage-plasticity in crystalline rock

# Crystal plasticity theory – Kinematics

- Decomposition of the Total Strain:

$$\boldsymbol{\epsilon} := \nabla^{\text{sym}} \boldsymbol{u} = \frac{1}{2} (\nabla \boldsymbol{u} + \nabla^T \boldsymbol{u}) \Rightarrow \boldsymbol{\epsilon} = \boldsymbol{\epsilon}^e + \boldsymbol{\epsilon}^\theta + \boldsymbol{\epsilon}^p \text{ with } \boldsymbol{\epsilon}^p = \sum_{\alpha} \gamma^{\alpha} \boldsymbol{S}^{\alpha}.$$

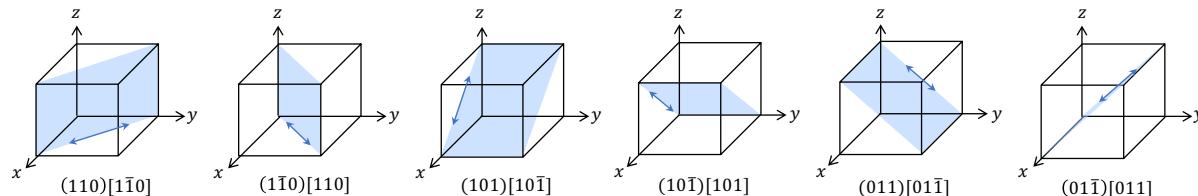
$\boldsymbol{\epsilon}^p$ : the integration over all the crystallographic slips

$\gamma^{\alpha}$ : the plastic slip corresponding to the slip-system  $\alpha$

$\boldsymbol{S}^{\alpha}$ : the symmetric part of the *Schmid Tensor* (i.e.  $\boldsymbol{s}^{\alpha} \otimes \boldsymbol{m}^{\alpha}$ )

$$\boldsymbol{S}^{\alpha} = \frac{1}{2} (\boldsymbol{m}^{\alpha} \otimes \boldsymbol{s}^{\alpha} + \boldsymbol{s}^{\alpha} \otimes \boldsymbol{m}^{\alpha})$$

- The Slip-System for Single-Crystal Halite  $\{110\}\langle1\bar{1}0\rangle$  (cf. FCC metal  $\{111\}\langle110\rangle$ )



## Crystal plasticity theory – Rate Dependence (Creep)

- The stress update algorithm for both rate-dependent and independent behavior
  - Flow rule (viscoplastic regularization)

$$\dot{\gamma}^\alpha = C_0 \exp\left(-\frac{Q}{RT}\right) \left[ \left( \frac{f^{\alpha+}}{\tau_Y^\alpha} + 1 \right)^p - 1 \right] \quad \text{with} \quad f^{\alpha+} := \begin{cases} f^\alpha & \text{if } f^\alpha > 0 \\ 0 & \text{otherwise} \end{cases}$$

where  $\beta$  indicates each slip system;  $C_0$  shape factor;  $Q$  activation energy;  $R$  the gas constant;  $T$  temperature;  $p$  exponent;  $f^\alpha$  yield condition;  $\tau_Y^\alpha$  the critical resolved shear stress.

- The residual and Jacobian matrix (cf. Miehe and Schörder, 2001)

$$r^\alpha = \boldsymbol{\sigma} : \boldsymbol{S}^\alpha - \tau_Y^\alpha \sqrt[p]{\frac{\eta}{\Delta t} \gamma^\alpha + 1} \quad \text{with} \quad \eta = 1 / \left[ C_0 \exp\left(-\frac{Q}{RT}\right) \right]$$
$$D^{\alpha\beta} = \boldsymbol{S}^\alpha : \mathbb{C}^e : \boldsymbol{S}^\beta + h \left( \frac{\eta}{\Delta t} \gamma^\alpha + 1 \right)^{1/p} + \tau_Y^\alpha \delta^{\alpha\beta} \frac{\eta}{p\Delta t} \left( \frac{\eta}{\Delta t} \gamma^\alpha + 1 \right)^{1/p-1}$$

- The singular value decomposition (SVD) and pseudo-inverse method are used to regularize the inverse of Jacobian matrix (also see Ultimate algorithm, Borja and Wren, IJNME, 1997).

# Anisotropic damage

- Modeling a single-crystal halite using thermomechanics:

- Coupled anisotropic inelasticity and fracture behavior.
- Rate and temperature dependence – dislocation creep.
- Intra- and intergranular fracture behavior.

- Governing equations

- Balance of linear momentum

$$\nabla \cdot \boldsymbol{\sigma} + \mathbf{b} = \mathbf{0} \quad \text{with} \quad \boldsymbol{\sigma} = \frac{\partial \hat{\psi}}{\partial \boldsymbol{\epsilon}^e} = g(\underline{d}) \hat{\boldsymbol{\sigma}} \quad \text{with} \quad \hat{\boldsymbol{\sigma}} = \mathbb{C}^e : \boldsymbol{\epsilon}^e - \alpha(\theta - \theta_0) \mathbf{1}$$

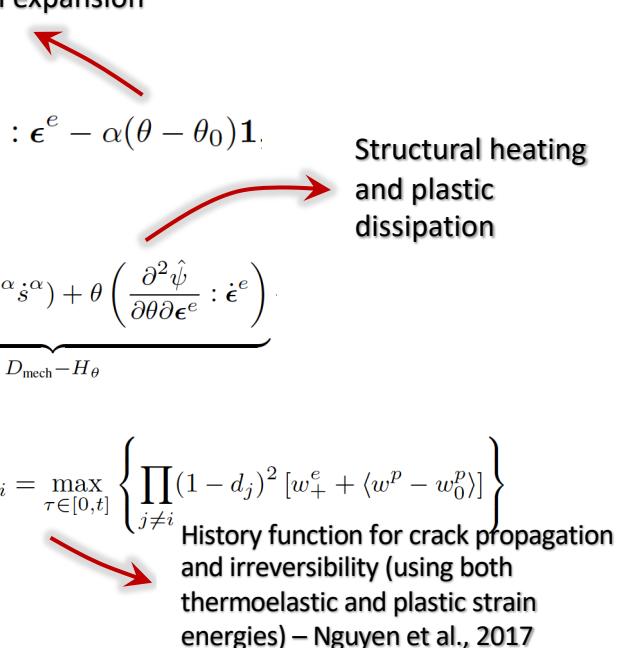
- Balance of energy

$$c_v \dot{\theta} = [D_{\text{mech}} - H_\theta] - \nabla \cdot \mathbf{q} + r_\theta \quad \text{where} \quad \underbrace{\sum_\alpha (\pi^\alpha \dot{\gamma}^\alpha + g^\alpha \dot{s}^\alpha) + \theta \left( \frac{\partial^2 \hat{\psi}}{\partial \theta \partial \boldsymbol{\epsilon}^e} : \dot{\boldsymbol{\epsilon}}^e \right)}_{D_{\text{mech}} - H_\theta}$$

- Multiple phase-field equation (microforce balance)

$$2(1 - d_i) \mathcal{H}_i + \frac{G_c}{l} d_i + G_c l \nabla \cdot (\boldsymbol{\omega}_i \cdot \nabla d_i) = 0 \quad \text{with} \quad \mathcal{H}_i = \max_{\tau \in [0, t]} \left\{ \prod_{j \neq i} (1 - d_j)^2 [w_+^e + \langle w^p - w_0^p \rangle] \right\}$$

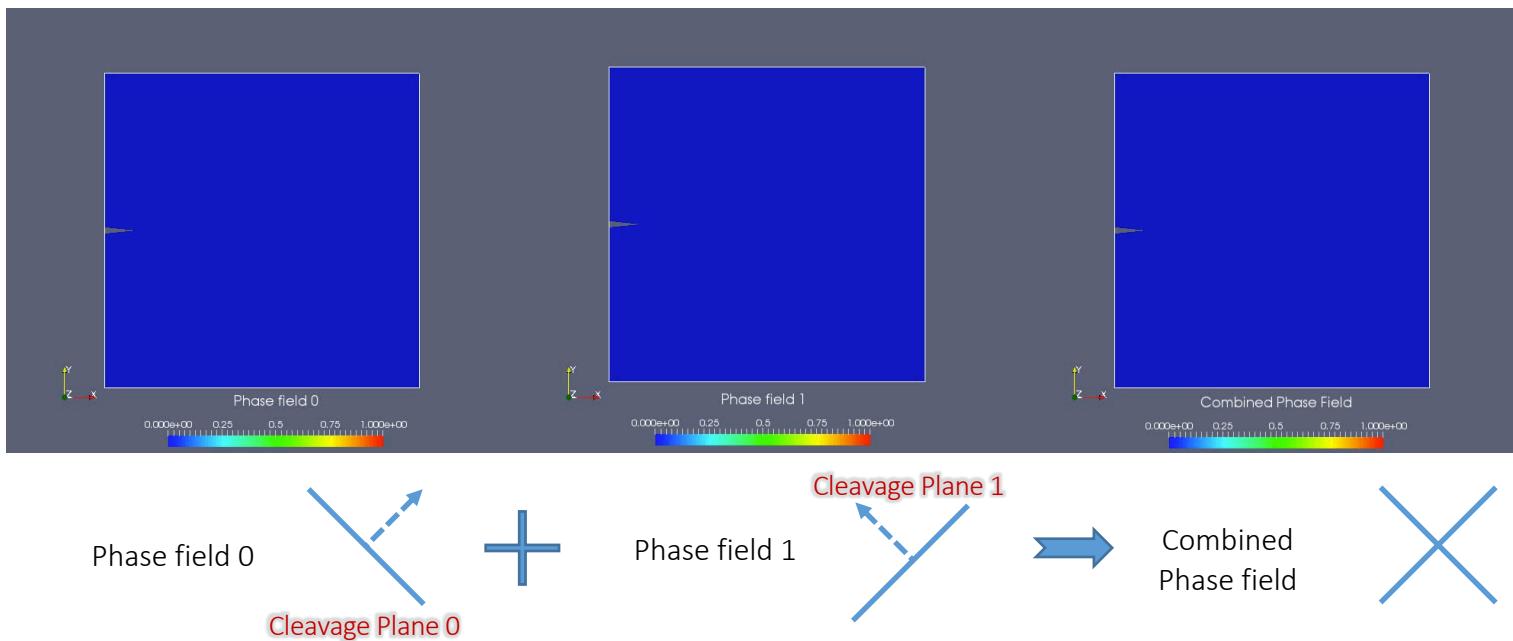
Second-order structural tensor(for directional fracture - Knap and Clayton, 2015 )



## Benchmarks

- **Why we use the multiple phase-field method? How does the method capture anisotropic fracture?**

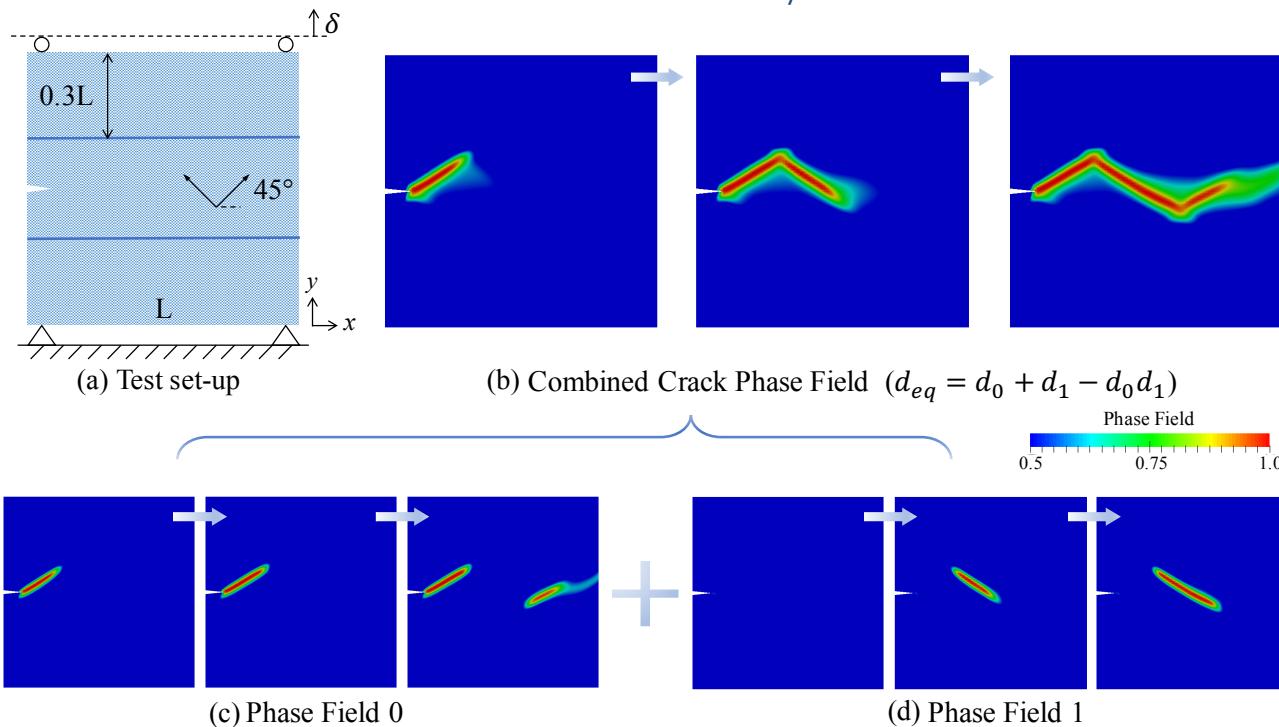
- The existing standard FEM code can be easily extended to capture anisotropic damage (without using the higher-order phase-field or constitutive equation).
- Example: guided crack propagation (cf. Li et al., 2015, Nguyen et al., 2017).



## Benchmarks

- Why we use the multiple phase-field method? How does the method capture anisotropic fracture?

- We can relate the number of the slip-system to the number of multiple phase-field variables.
- This method can be further utilized to model recrystallization.



# Formulations

- Modeling a single-crystal halite using thermomechanics:

- Anisotropic inelastic and fracture behavior.
- Rate and temperature dependence – dislocation creep.
- Intra- and intergranular fracture behavior.

- Governing equations

- Balance of linear momentum

$$\nabla \cdot \boldsymbol{\sigma} + \mathbf{b} = \mathbf{0} \quad \text{with} \quad \boldsymbol{\sigma} = \frac{\partial \hat{\psi}}{\partial \boldsymbol{\epsilon}^e} = g(\underline{d}) \hat{\boldsymbol{\sigma}} \quad \text{with} \quad \hat{\boldsymbol{\sigma}} = \mathbb{C}^e : \boldsymbol{\epsilon}^e - \alpha(\theta - \theta_0) \mathbf{1}$$

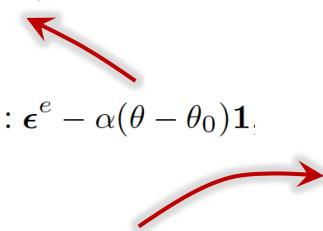
- Balance of energy

$$c_v \dot{\theta} = [D_{\text{mech}} - H_\theta] - \nabla \cdot \mathbf{q} + r_\theta \quad \text{where} \quad \underbrace{\sum_\alpha (\pi^\alpha \dot{\gamma}^\alpha + g^\alpha \dot{s}^\alpha) + \theta \left( \frac{\partial^2 \hat{\psi}}{\partial \theta \partial \boldsymbol{\epsilon}^e} : \dot{\boldsymbol{\epsilon}}^e \right)}_{D_{\text{mech}} - H_\theta}$$

- Multiple phase-field equation (micro-force balance) (cf. Gurtin 1996, Na & Sun, 2017, Choo & Sun, 2018a,b).

$$2(1 - d_i) \mathcal{H}_i + \frac{G_c}{l} d_i + G_c l \nabla \cdot (\boldsymbol{\omega}_i \cdot \nabla d_i) = 0 \quad \text{with} \quad \mathcal{H}_i = \max_{\tau \in [0, t]} \left\{ \prod_{j \neq i} (1 - d_j)^2 [w_+^e + \langle w^p - w_0^p \rangle] \right\}$$

a second-order structural tensor  
(for directional fracture)



Structural heating  
and plastic  
dissipation

History function for crack propagation  
and irreversibility (using both  
thermoelastic and plastic strain energies)

# Formulations

- How is each balance equation related?

- A combined stored energy functional

$$\psi = \hat{\psi}^e(\epsilon^e, \theta, \underline{d}) + \hat{\psi}^p(\underline{s}, \underline{d}) + \hat{\psi}^c(\underline{d}, \nabla \underline{d}, \omega) + \hat{\psi}^\theta(\theta)$$

**Thermoelastic strain energy**

$$\hat{\psi}^e(\epsilon^e, \theta, \underline{d}) = g(\underline{d}) w^e(\epsilon^e, \theta) \quad \text{with} \quad w^e(\epsilon^e, \theta) = \frac{1}{2} \epsilon^e : \mathbb{C}^e : \epsilon^e - \alpha(\theta - \theta_0) \operatorname{tr} \epsilon^e$$

**Plastic strain energy**

$$\hat{\psi}^p(\underline{s}, \underline{d}) = g(\underline{d})^p \langle w^p - w_0^p \rangle, \quad \text{with } g(\underline{d})^p = g(\underline{d}) \text{ and } w^p = \frac{1}{2} \sum_{\alpha} h(s^{\alpha})^2$$

**Total crack surface energy**

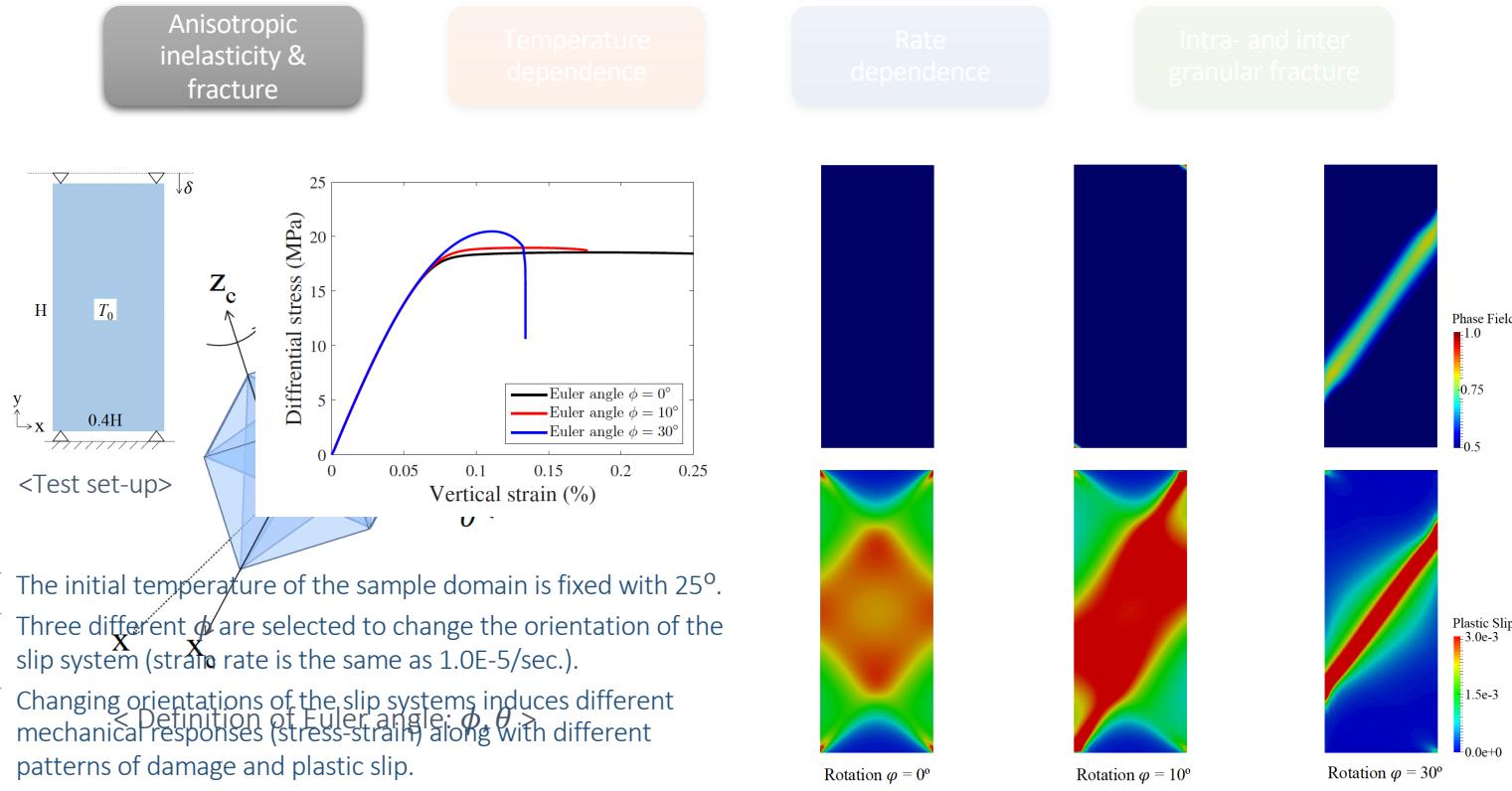
$$\hat{\psi}^c = G_c \sum_i \left[ \frac{1}{2l} (d_i)^2 + \frac{l}{2} \omega_i : (\nabla d_i \otimes \nabla d_i) \right]$$

**Purely thermal energy for heat transfer**

$$\hat{\psi}^\theta = c_v [(\theta - \theta_0) - \theta \ln(\theta/\theta_0)]$$

# Applications

- Modeling a single-crystal halite using thermomechanics:



- ✓ The initial temperature of the sample domain is fixed with  $25^\circ$ .
- ✓ Three different  $\phi$  are selected to change the orientation of the slip system (strain rate is the same as  $1.0E-5/\text{sec.}$ ).
- ✓ Changing orientations of the slip systems induces different mechanical responses (stress-strain) along with different patterns of damage and plastic slip.

# Applications

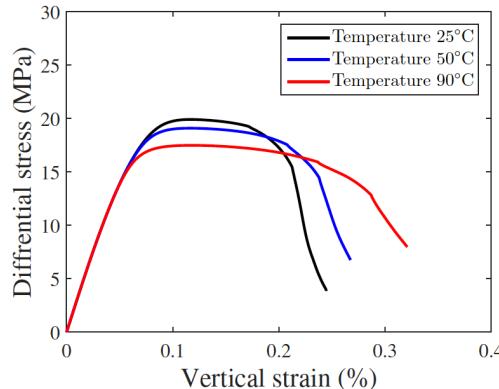
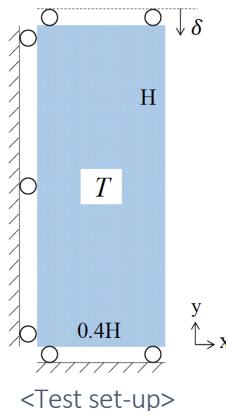
## ○ Modeling a single-crystal halite using thermomechanics:

Anisotropic  
inelasticity &  
fracture

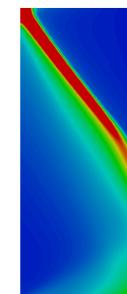
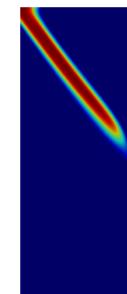
Temperature  
dependence

Rate  
dependence

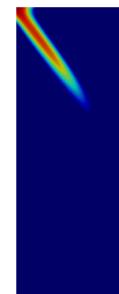
Intra- and inter  
granular fracture



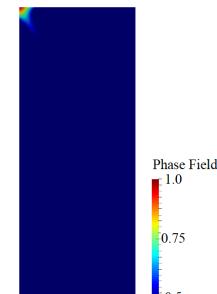
- ✓ Three initial temperatures of the sample domain are selected to analyze the thermal impact on mechanical responses.
- ✓ As the temperature increases, the more ductile behavior is obtained from stress-strain curves.
- ✓ The thermal dissipation suppresses the crack propagation and leads to different plastic patterns.



Temperature 25°C



Temperature 50°C



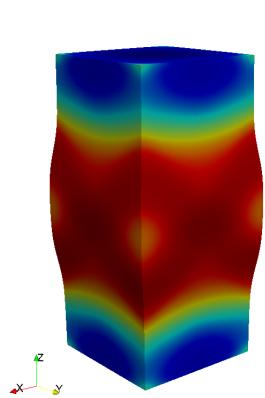
Temperature 90°C

Temperature 90°C

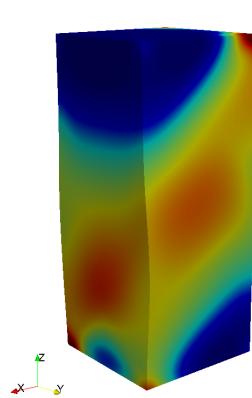
Note: The results are captured at the same vertical strain level (0.24%), and Euler angles are fixed for all cases:  $\theta = 0^\circ$  and  $\phi = 70^\circ$

# Modeling a single-crystal halite in 3D

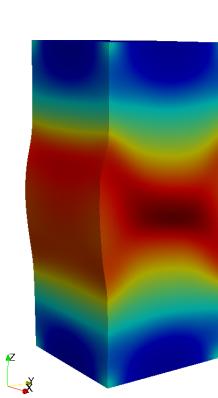
- All the slip-systems associated with each crack phase-field (different orientations )



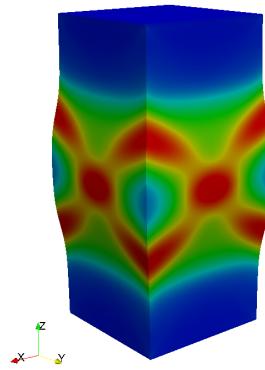
$$\theta = 0^\circ \\ \phi = 0^\circ$$



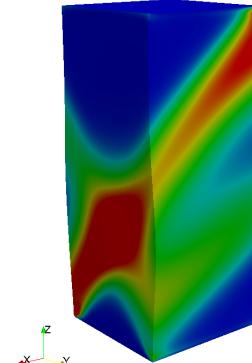
$$\theta = 30^\circ \\ \phi = 10^\circ$$



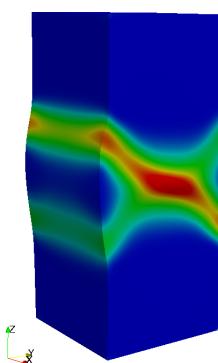
$$\theta = 45^\circ \\ \phi = 0^\circ$$



$$\text{Plastic Slip}$$



$$\text{Plastic Slip}$$



$$\text{Plastic Slip}$$

# Strongly Anisotropic materials in the brittle regime

# FFT-based High-order Phase-field

Regularized cleavage surface energy for multi-phase-field

$$\gamma(d, \nabla d, \nabla^2 d) = \frac{1}{2l_0}d^2 + \frac{l_0}{4}\nabla d \cdot \nabla d + \frac{l_0^3}{32}\nabla^2 d : \mathbb{A} : \nabla^2 d + \frac{1}{2}\beta_1(d-1)^2$$

Cleavage energy      Initial phase field  
anisotropy      penalization

First order variation, integration by part twice

$$\int_{\Omega} \left[ 2(d-1)\mathcal{H}_1^1 + \frac{G_c}{l_0}d - \nabla \cdot \left( \frac{G_c l_0}{2}\nabla d \right) + \nabla^2 : \left( \frac{G_c l_0^3}{16}\mathbb{A} : \nabla^2 d \right) + \beta_1(d-1) \right] \delta dd\Omega$$

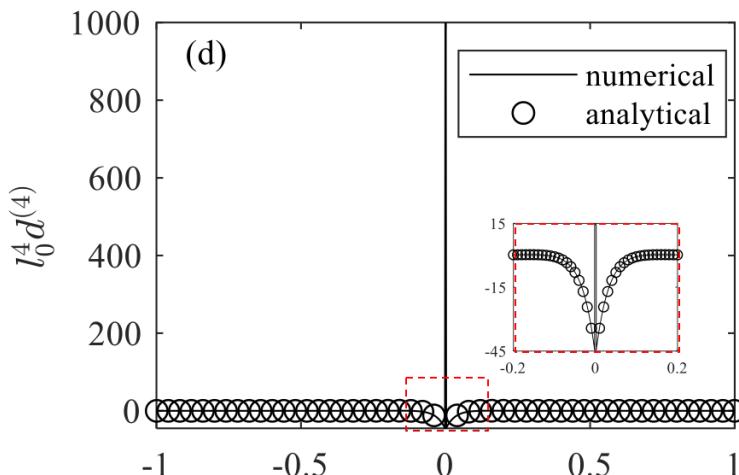
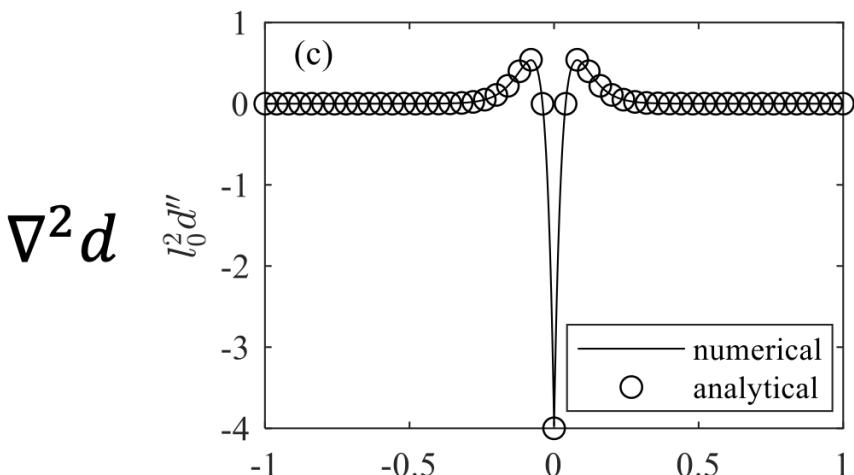
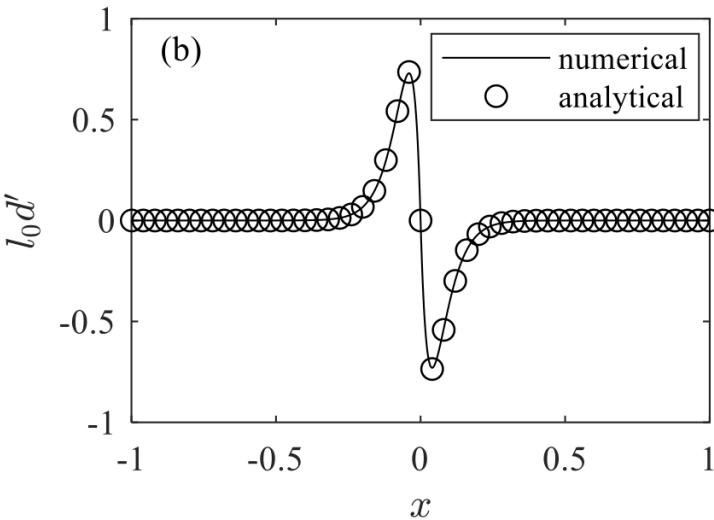
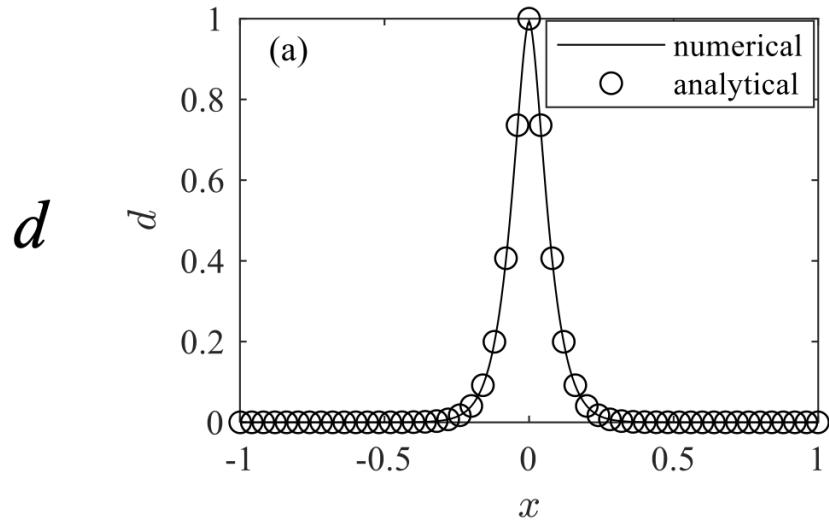
$$+ \underbrace{\int_{\partial\Omega} \left[ \frac{G_c l_0}{2}\nabla d \cdot \vec{n} + \frac{G_c l_0^3}{16}\nabla^2 d : \mathbb{A} : (\nabla \delta d \otimes \vec{n}) - \frac{G_c l_0^3}{16}\nabla \cdot (\mathbb{A} : \nabla^2 d) \cdot \vec{n} \delta d \right] dS}_{\text{Surface integral over periodic RVE}} = 0.$$

**Assumptions:**

- 1)  $A, G_c$  and  $l_0$  are periodic  $\rightarrow$  **surface integral** vanishes
- 2)  $A, G_c$  are piecewise constant,  $l_0$  is constant  $\rightarrow \text{grad}(A) = 0$

$$\left( 2\psi_1^1 + \frac{G_c}{l_0} + \frac{G_c l_0}{2}\mathcal{F}^{-1}\boldsymbol{\xi} \cdot \boldsymbol{\xi}\mathcal{F} + \frac{G_c l_0^3}{16}\mathbb{A} : \mathcal{F}^{-1}\boldsymbol{\xi} \otimes \boldsymbol{\xi} \otimes \boldsymbol{\xi} \otimes \boldsymbol{\xi}\mathcal{F} + \beta \right) d = 2\psi_1^1 + \beta.$$

# Numerical Examples (1D rod)



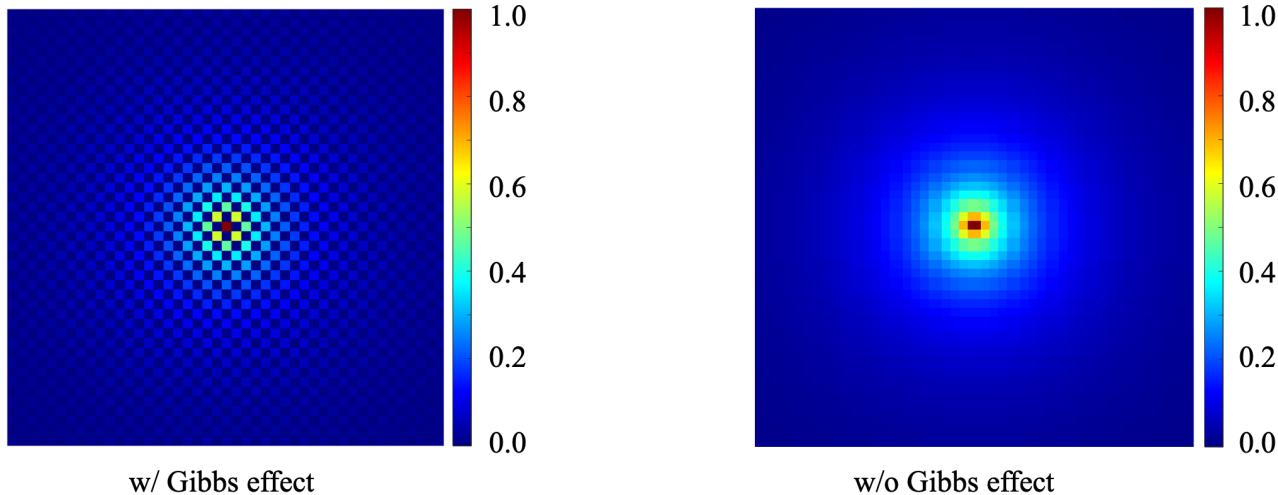
$\nabla d$

$\nabla^4 d$

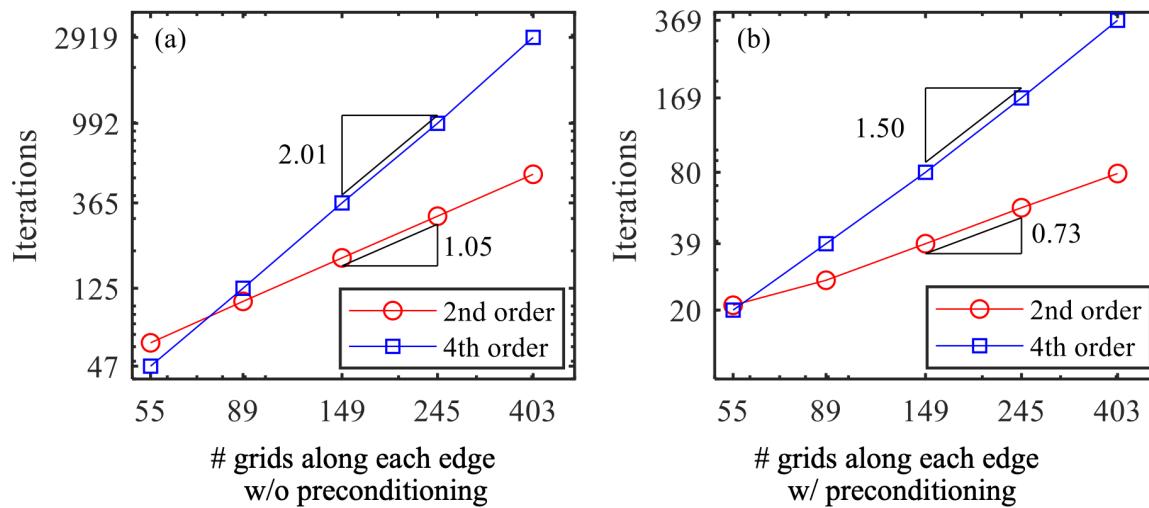
- trigonometric polynomials accurately calculate high order gradient without spurious oscillations
- Gibbs effect is dampened by using finite difference based frequency vectors

# Numerical Accuracy and Efficiency of FFT-based method

## Gibbs effect



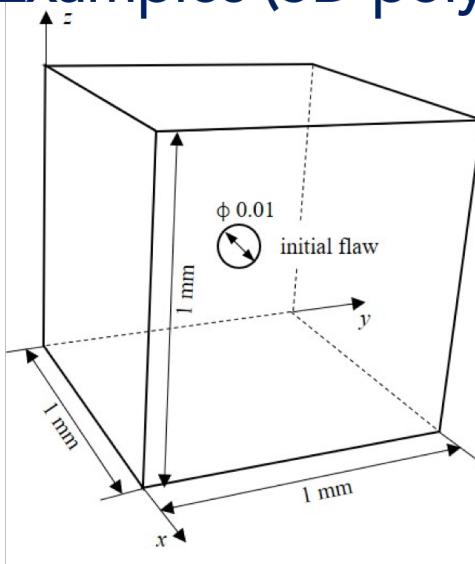
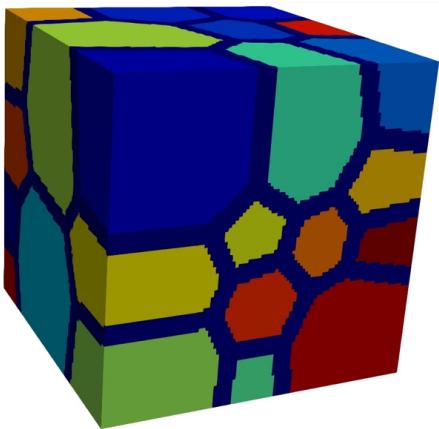
## Numerical Efficiency



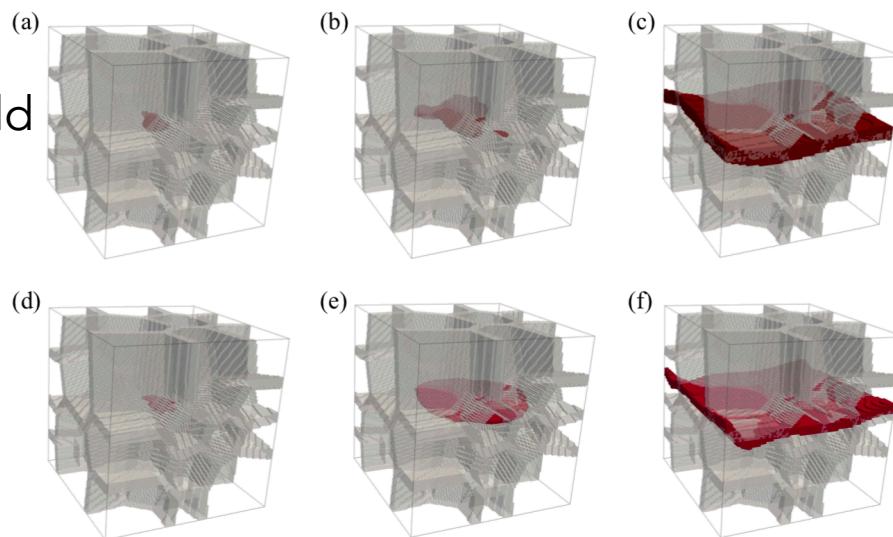
Observation:

- high-order phase-field more expensive than multi-phase-field
- preconditioning matrix is effective in accelerating the iterative linear solver

# Numerical Examples (3D polycrystalline rock salt)



Multi-phase-field  
(MPM)



High-order  
phase-field  
(HPM)

Material:

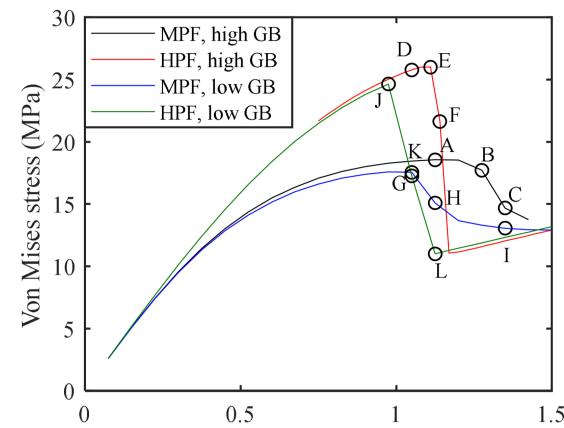
- 3D polycrystalline rock salt
- 40 grains with random initial orientation
- $101^3$  Fourier grid points

Load case:

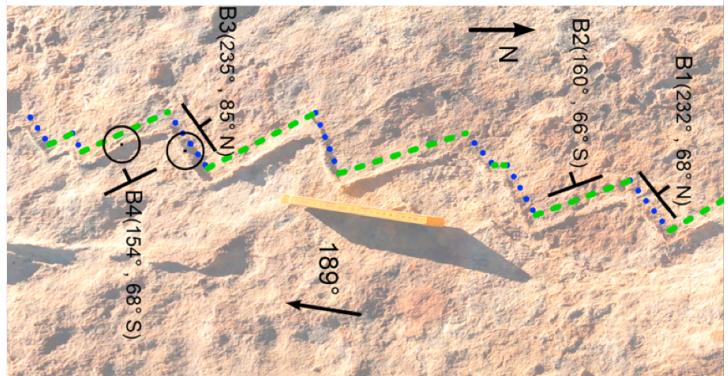
- Uniaxial tension of RVE with central flaw

Constitutive:

- mechanical: small strain elasticity
- phase field: strongly anisotropy



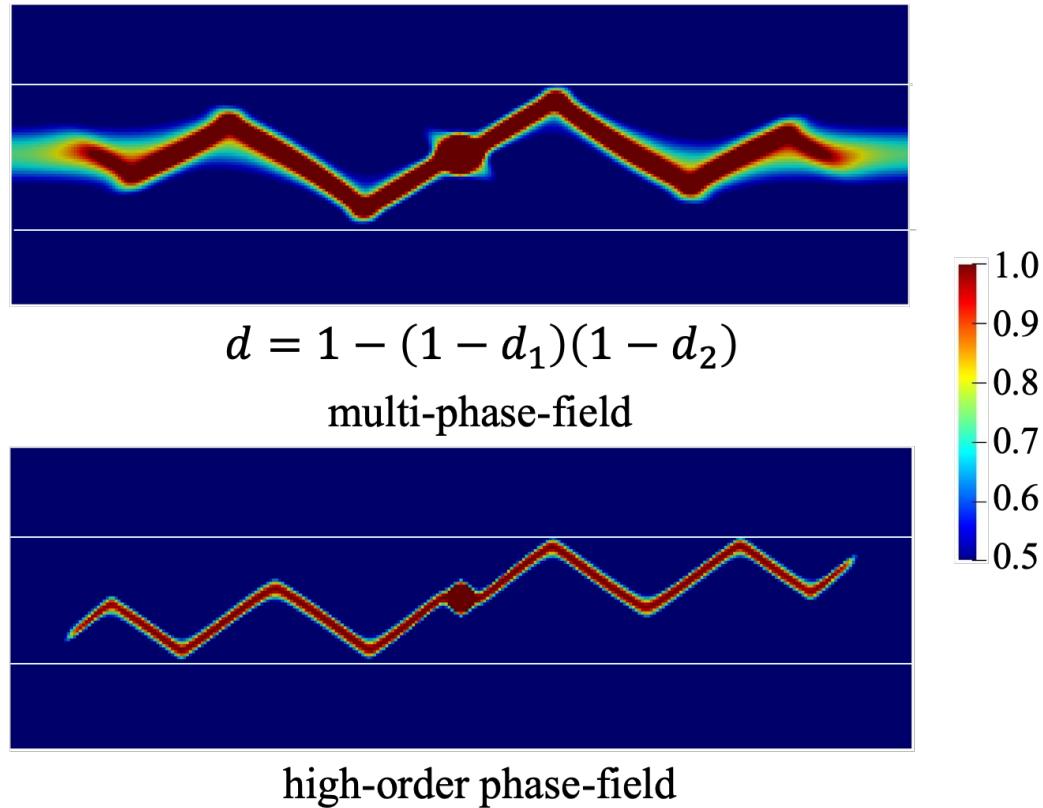
# Sawtooth crack pattern in strongly anisotropic phase field fracture model



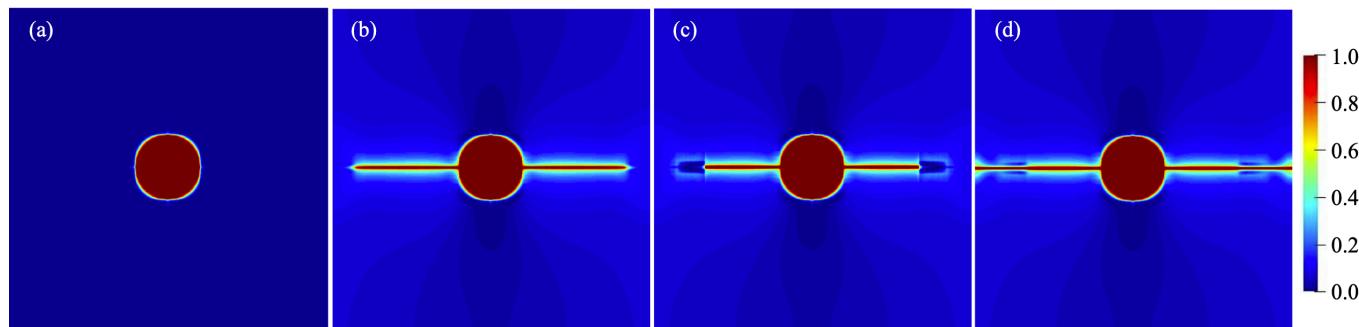
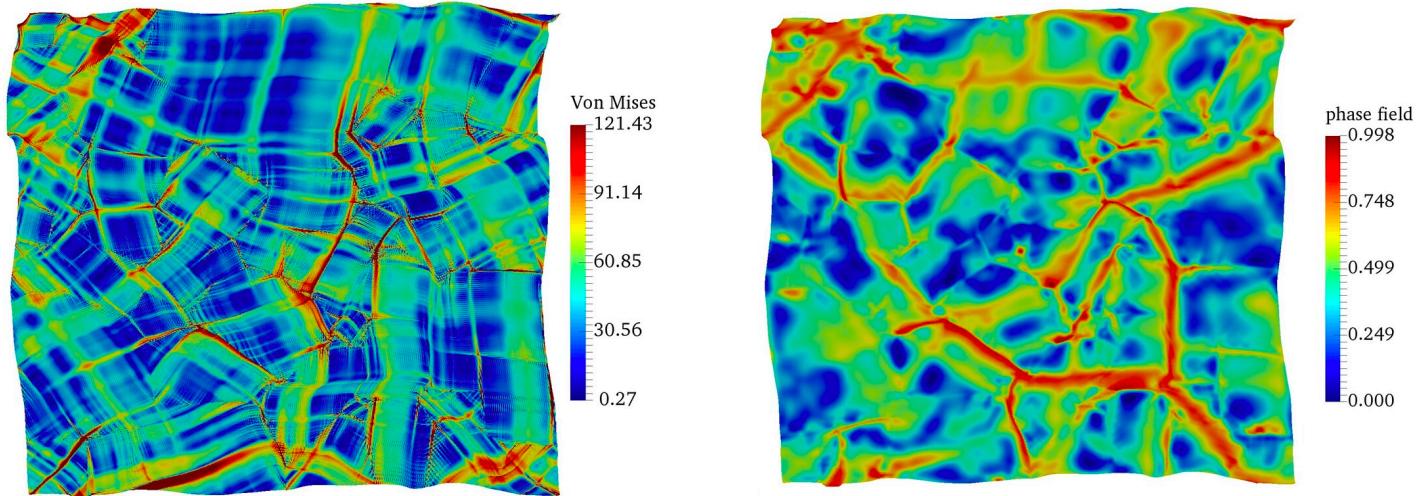
<Outcrop of wiggly compaction bands (from Liu et al., 2015)>

## Observations

- Both multi-phase-field and high-order phase-field can capture the sawtooth crack pattern
- Multi-phase-field is more regularized
- Difference between MPF and HPF is caused by numerical parameters

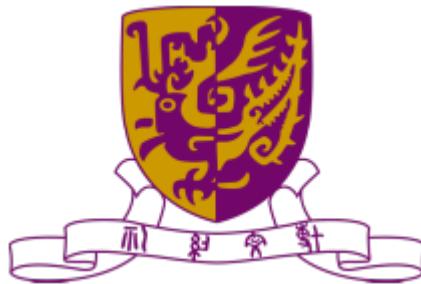


# Fracture healing and precipitating creeping



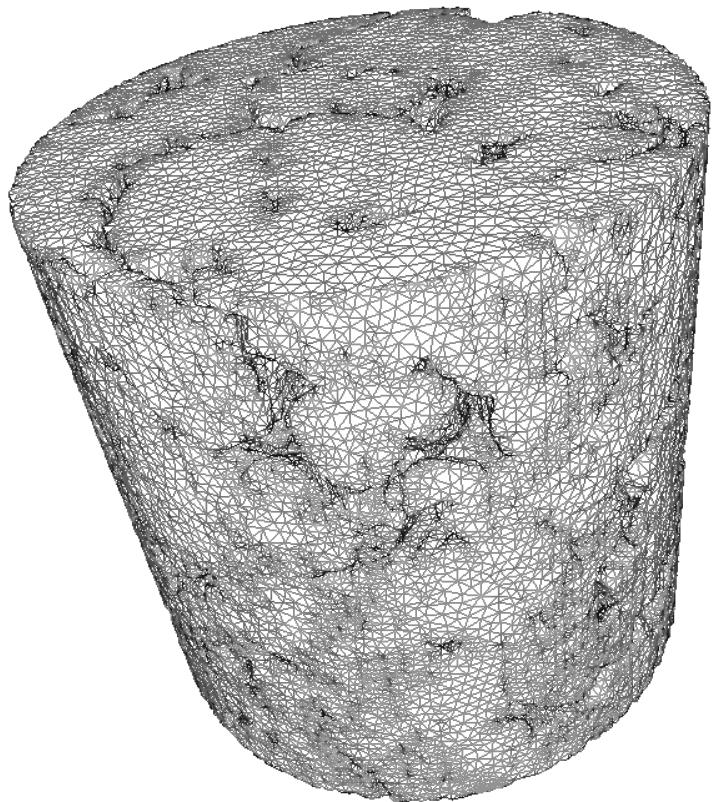
# Acknowledgements

- Army Research Office: Young Investigator Program Award
- Air Force Office of Scientific Research, Young Investigator Program Award
- National Science Foundation: EAR-1516300 and CMMI-1445033
- Sandia National Laboratories
- US Department of Energy Office of Nuclear Energy
- Columbia University





COLUMBIA UNIVERSITY  
IN THE CITY OF NEW YORK



**THANK YOU!**  
More information can be found at  
[www.poromechanics.org](http://www.poromechanics.org)